

Survey of Theories of Glassy Behavior

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Outline of the talk

- What is a theory?
- Phenomenological descriptions
 - Structure
 - Continuous Random Network
 - Random close packing
 - Dynamics
 - Adam Gibbs theory
 - Landscapes
 - Free volume theory
- Microscopic descriptions
 - Rigidity percolation
 - Mode Coupling Theory (MCT)
 - Random first order theory (RFOT)

What is a theory?

- Distinguish fitting function and theory!
 - Fitting functions:
 - Kohlrausch-Williams-Watts function
 - Coupling model (K. Ngai)
 -
 - Theory: Should allow to make a *microscopic* calculation for a given Hamiltonian; calculations might be difficult and might be approximate; results might be bad

N.B.'s:

1) There are complicated models, e.g. kinetically facilitated Ising models (see talk by Jack), that allow to reproduce a few dynamic aspects of real glass-forming liquids; these models are useful to understand certain mechanisms, but they are models and not theories

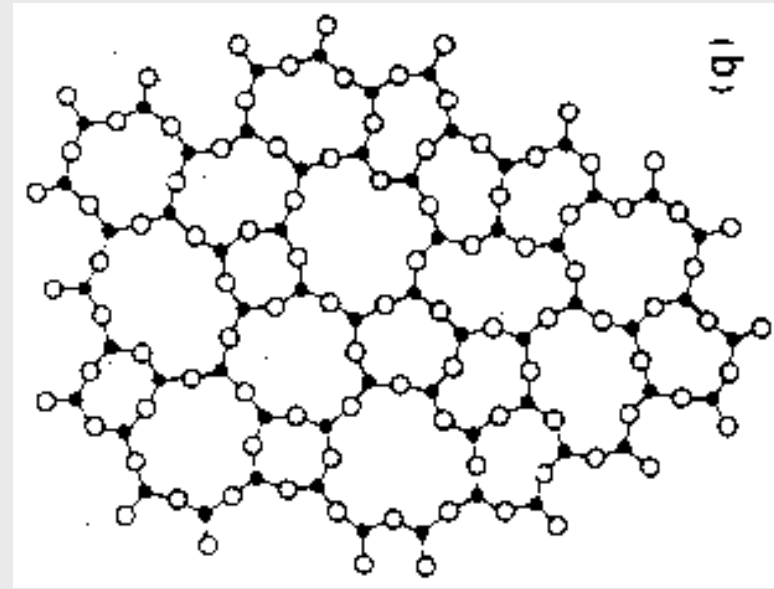
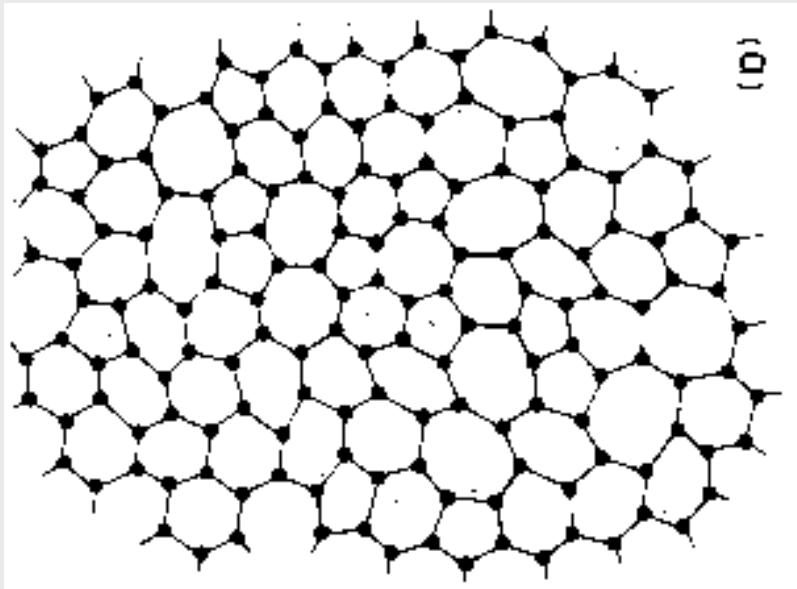
2) In glass physics the sophistication of approaches/theories spans orders of magnitudes!

Phenomenological description of the structure: Continuous Random Network

- Covalently bond atoms have a very well defined coordination number
⇒ well defined local chemical order; Si: $z=4$, C: $z=4$; As: $z=3$,...
- But angles can still vary ⇒ local geometrical disorder ⇒ random network
- Simple rule to build a random network

For example $d=2$ random network

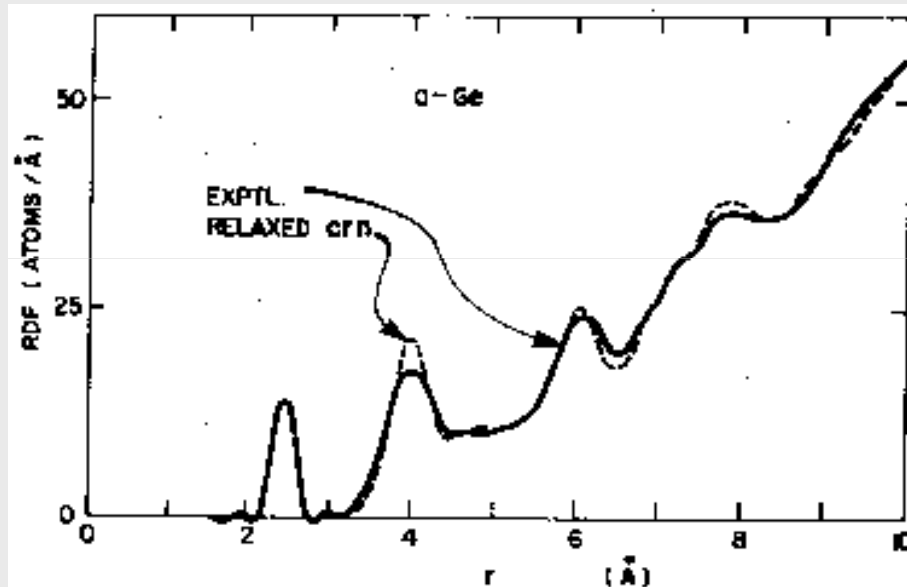
- all atoms have $z=3$
- all nearest neighbor distances are fixed
- no loose ends (=dangling bonds)



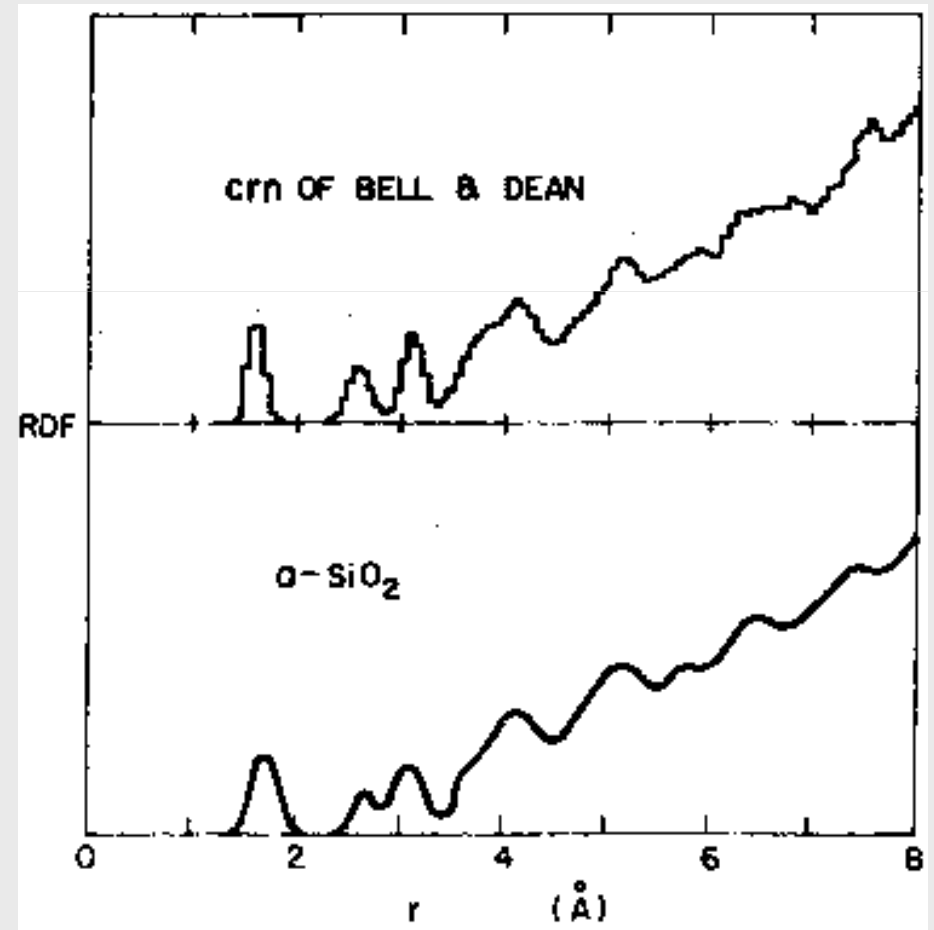
Phenomenological description of the structure: Continuous Random Network: 2

- Comparing structure from CRN with experiments

Amorphous Ge
(Steinhardt et al. 1974)



Amorphous SiO₂
(Bell and Dean 1972)



Phenomenological description of the structure: Random Close Packing

- **Approximate atoms as hard spheres**; this is ok for many metals (see talks by Kelton and Poole)
- **Dimension $d=2$** : Tight packing of hard disks is simple \Rightarrow **hexagonal lattice**
- **Dimension $d=3$** : Local best packing is a **tetrahedron**; BUT tetrahedra do not allow to fill space uniformly \Rightarrow make local compromise and use **icosahedron**; BUT icosahedra do not allow to fill space uniformly \Rightarrow **random structure** (=glass); (see talk by Wales)

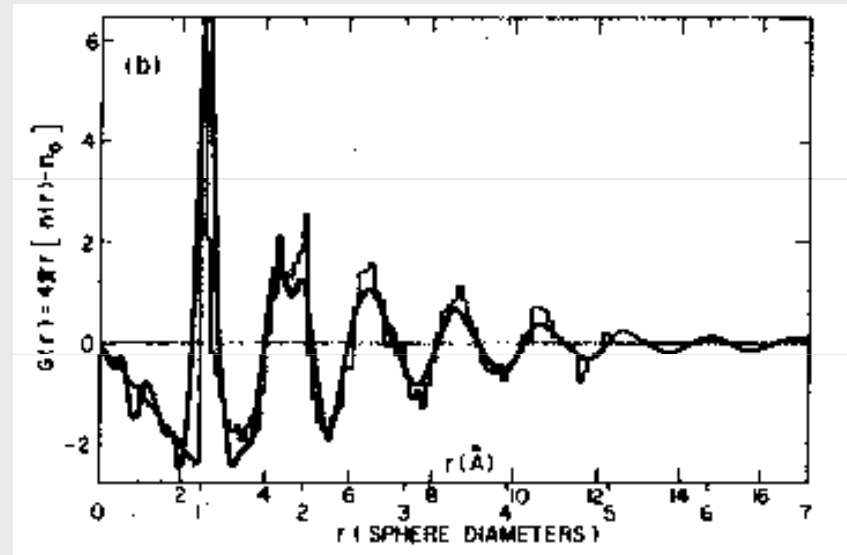
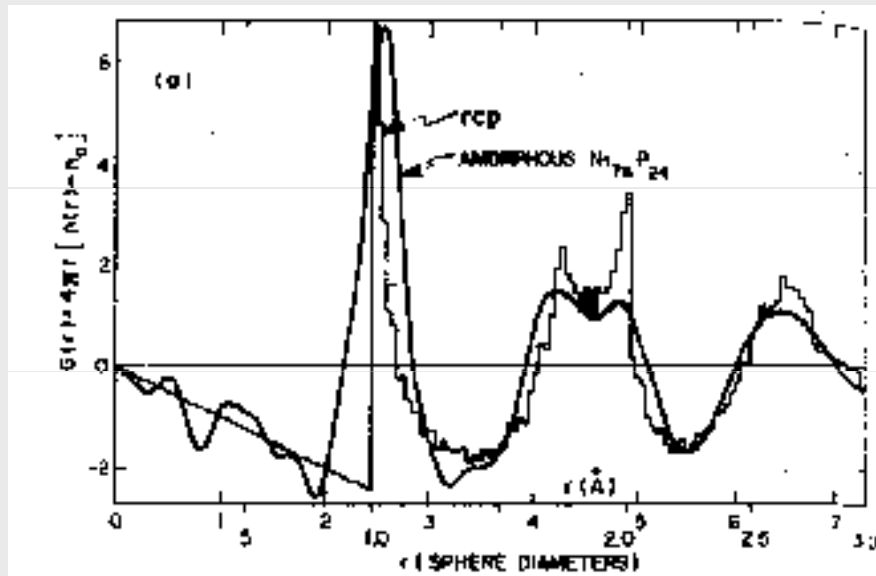
NB: **The structure of random close packing is not unique** (location of the particles and average density) but depends on the procedure how the structure was created!

- Typical value of the density of RCP: $\rho_{\max} \approx 0.64$; compare with tight crystalline packing with : $\rho_{\max} \approx 0.74$ (HCP or FCC structure)

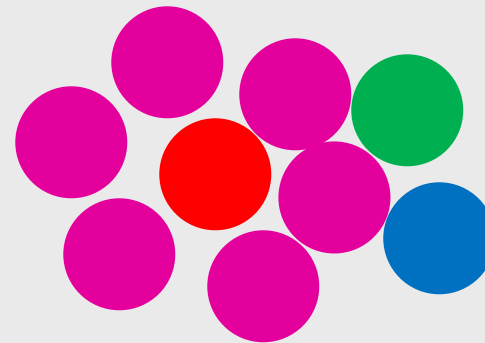
Phenomenological description of the structure: Random Close Packing: 2

- Comparing structure from RCP with experiments:

amorphous $\text{Ni}_{76}\text{P}_{24}$ (Cargill 1975)



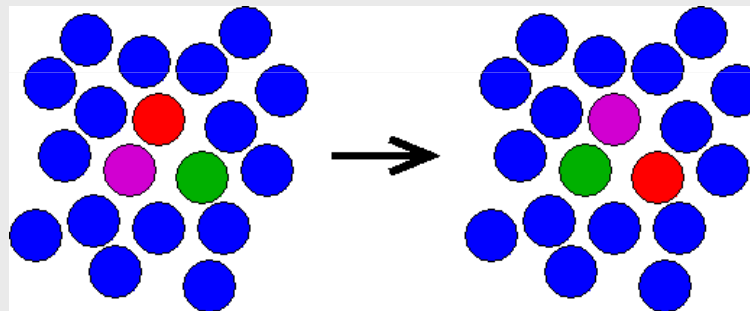
- Reasonably good reproduction of the structure
- Splitting of the second nearest neighbor peak



The theory of Adam and Gibbs

Basic idea: (Adam and Gibbs 1965)

At low T the relaxation dynamics is a sequence of individual events in which a subregion of the liquid relaxes to a new local configuration. These rearrangements are not single particle jumps (like in a crystal) but cooperative
⇒ Cooperatively rearranging regions (CRR)



Assumptions:

- The CRRs are independent of each other
- The CRRs contain sufficiently many particles to allow to apply the formalism of statistical mechanics

The theory of Adam and Gibbs: 2

Consider one CRR that has z particles; the **isothermal-isobaric partition function for this subsystem** is then

$$\Delta(z, P, T) = \sum_{E, V} w(z, E, V) \exp(-\beta H)$$

where $w(z, E, V)$ is the number of states of the CRR with energy E and volume V , and H is the enthalpy

Not all allowed states can undergo a rearrangement!

⇒ introduce a partition function $\Delta'(z, P, T)$ that considers only the states that can undergo rearrangements

⇒ the fraction of systems that can undergo a relaxation event is

$$f(z, T) = \Delta' / \Delta = \exp(-\beta (G - G')) = \exp(-\beta z \delta\mu)$$

with the Gibbs free energies $G = -k_B T \ln \Delta$ and $G' = -k_B T \ln \Delta'$, and $\delta\mu$ the difference in the chemical potential per particle.

$W(z, T)$, the probability that the system makes a cooperative rearrangement, is proportional to $f(z, T)$ and thus

$$W(z, T) = A \exp(-\beta z \delta\mu)$$

The theory of Adam and Gibbs: 3

Assume that the total system is composed of a collection of $n(k, T)$ CRRs with k particles ($k=1,2,\dots$). The average probability that a particle makes a rearrangement is then

$$W^*(T) = N^{-1} \sum_{z=z^*}^N z n(z, T) W(z, T)$$

where z^* corresponds to the smallest cluster that is able to rearrange (with $z^* > 1$) and N is the number of particles in the system.

$$\Rightarrow W^*(T) = \frac{z^* n(z^*, T) \exp[-\beta z^* \delta \mu]}{N} \sum_{z=z^*}^N \frac{z n(z, T)}{z^* n(z^*, T)} A \exp[-\beta (z - z^*) \delta \mu]$$

or, for $\beta \delta \mu \gg 1$,

$$W^*(T) = A' \exp(-\beta z^* \delta \mu)$$

\Rightarrow The CRRs relevant for the relaxation dynamics have size z^* since the larger ones are slower by a factor of $O(\exp(-\beta \delta \mu)) \ll 1$.

The theory of Adam and Gibbs: 4

What is the value of z^* ? At low T we can decompose the dynamics of the particles in vibrations around local minima and transitions between these minima (idea of Goldstein).

⇒ The **partition function** can be **factorized into two factors** (contribution from vibrations × number of minima with a given energy)

⇒ The total **entropy of the system can be written as a sum of the vibrational entropy, S_{vib} , + configurational entropy S_{conf}**

The number of CRRs in a system with N particles is $n(z^*, T) = N/z^*$. Each CRR has thus a configurational entropy $s_{conf} = S_{conf} / n(z^*, T)$

$$\Rightarrow z^* = N / n(z^*, T) = N s_{conf} / S_{conf}$$

With $W^*(T) = A' \exp(-\beta z^* \delta\mu)$ one thus obtains

$$W^*(T) = A' \exp \left[-\frac{\beta N s_{conf} \delta\mu}{S_{conf}} \right] = A' \exp \left[-\frac{C}{T S_{conf}} \right]$$

and assuming that the **relaxation time $\tau(T)$ is proportional to $W^*(T)^{-1}$:**

$$\tau(T) \propto \eta(T) \propto \exp \left[\frac{C}{T S_{conf}} \right]$$

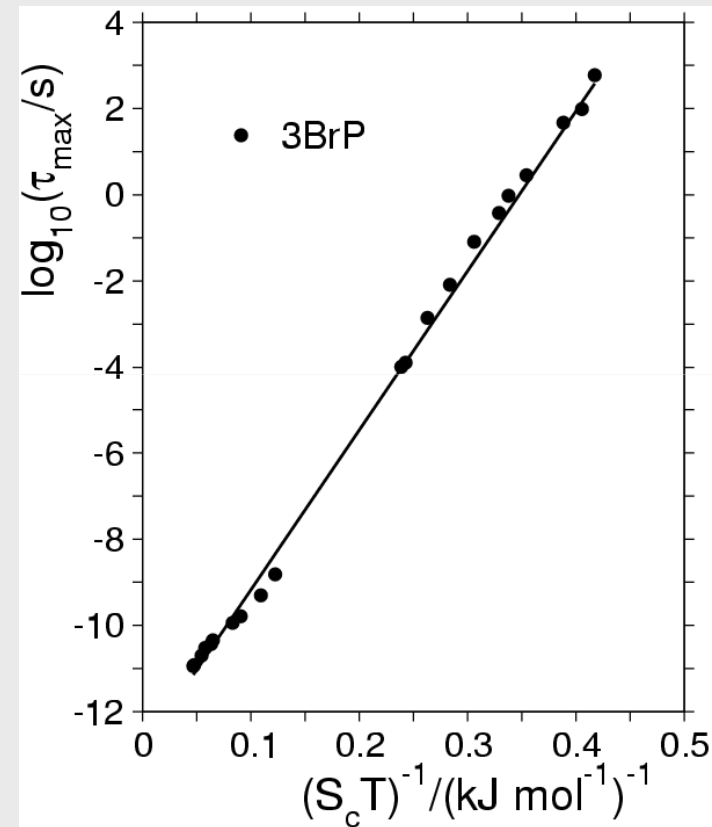
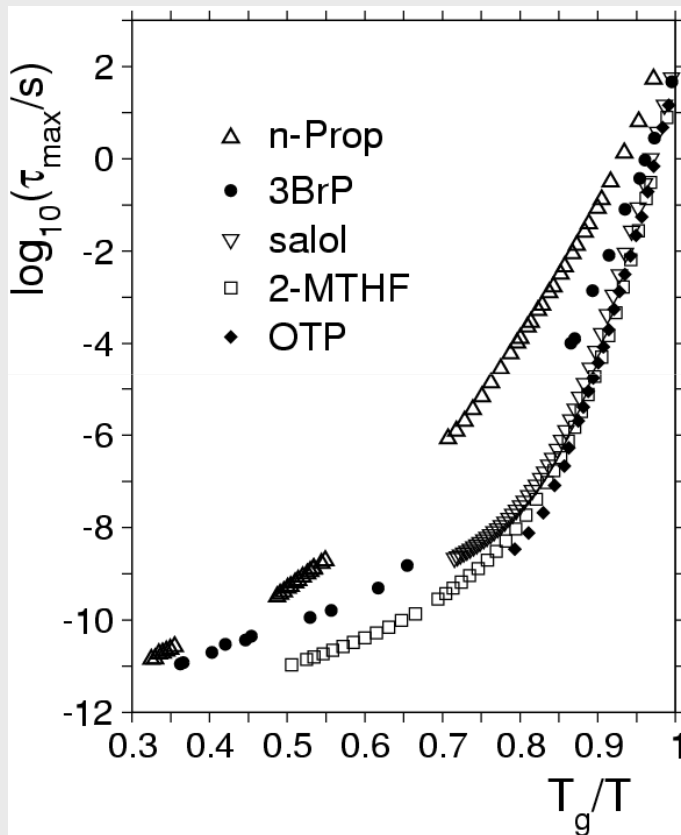
Relation of Adam-Gibbs

The theory of Adam and Gibbs: Validity

One can show that S_{conf} can be determined from the specific heat (Kauzmann)

⇒ The AG-relation can be tested experimentally

$$\tau(T) \propto \eta(T) \propto \exp \left[\frac{C}{TS_{\text{conf}}} \right]$$



Richert and Angell (1998) ⇒ AG works well over a large T-and τ -range (NB: No fit parameter!)

The theory of Adam and Gibbs: Consequences

In several glass-forming liquids the excess specific heat $\Delta C_p(T)$ (= spec. heat of liquid – spec. heat of crystal) can be fitted well by

$$\Delta C_p(T) = K/T$$

where K is a constant.

$$\Rightarrow \Delta S(T) = K (1/T_K - 1/T)$$

If we identify $\Delta S(T)$ with $S_{\text{conf}}(T)$ we obtain from the AG-relation:

$$\tau(T) \propto \exp \left[\frac{CT_K/K}{T - T_K} \right]$$

\Rightarrow The AG-relation is able to make a connection between dynamics and thermodynamics and to rationalize the Vogel-Fulcher law

Drawbacks of the AG-theory:

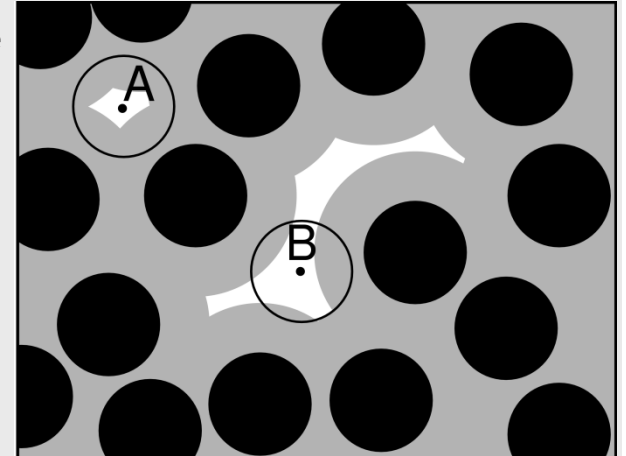
- What are the CRRs microscopically???
- Are the CRRs really independent?
- Is it reasonable to assume only one kind of CRRs?
- No predictions for other observables
- ...

Free Volume Theory

Cohen and Turnbull 1959-1970: Idea: A particle can only change its neighborhood if there is space to do so \Rightarrow need “free volume”

$$V_f = V - V_0$$

v = volume per particle; v_0 = volume (per particle) accessible only to one particle at a time (= volume of sphere)



Within **mean field** (= neglect correlations between adjacent free volumes) one can show easily that the probability to find a given free volume v' is given by

$$p(v') = \gamma / v_f \exp(-\gamma v' / v_f) \quad [\gamma = \text{geometric factor}]$$

For a **diffusing particle** that makes steps of size a with prob. $\Pi(a)$ one can show that the **diffusion constant** is prop. to $D \propto \int a \Pi(a) da$

\Rightarrow With $\Pi(a)=p(a)$ one obtains

$$D \propto \exp(-C / v_f)$$

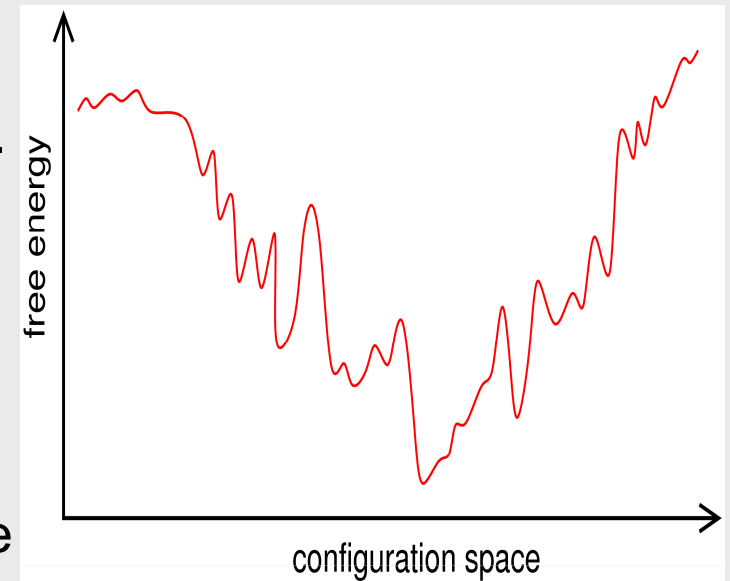
Thus $D \rightarrow 0$ if $v_f \rightarrow 0$

Thermal systems: $v_f = \alpha(T - T_0) \Rightarrow$ **Vogel-Fulcher law!**

Landscapes (see talks Sastry and Wales)

Goldstein 1969: The dynamics of the liquid becomes sluggish because the system has to overcome local barriers in configuration space. With decreasing T these barriers increase \Rightarrow slowing down is faster than Arrhenius

\Rightarrow **At low T the system is moving in a very rough landscape.** This can be shown to be exact in mean field spin glasses (see talks by Dasgupta and Mezard). NB: These barriers are in free energy!



What are the properties of this landscape? How many minima at what height? Distribution of barrier height. Is there a hierarchy in the landscape?

One can make models of landscapes (e.g. random energy model by Derrida 1980, trap model by Bouchaud et al. 1996) and impose a dynamics \Rightarrow complex relaxation that shows glassy features

Rigidity Percolation (see talk Elliott)

- **Phillips, Thorpe, Boolchand (1974--)**: Idea: A structure of (many) joints and stiff bars becomes rigid if the **number of constraints, n_c** , equals the **number of degrees of freedom, n_d** :

$$n_c = n_d$$

Consider a glass of N particles with n_r particles having coordination number r ($r = 1, \dots$); example $\text{Ge}_x\text{S}_{1-x-y}\text{I}_y$; $r = 4, 2$, and 1

A counting argument shows that the number of **floppy modes** (per particle) is

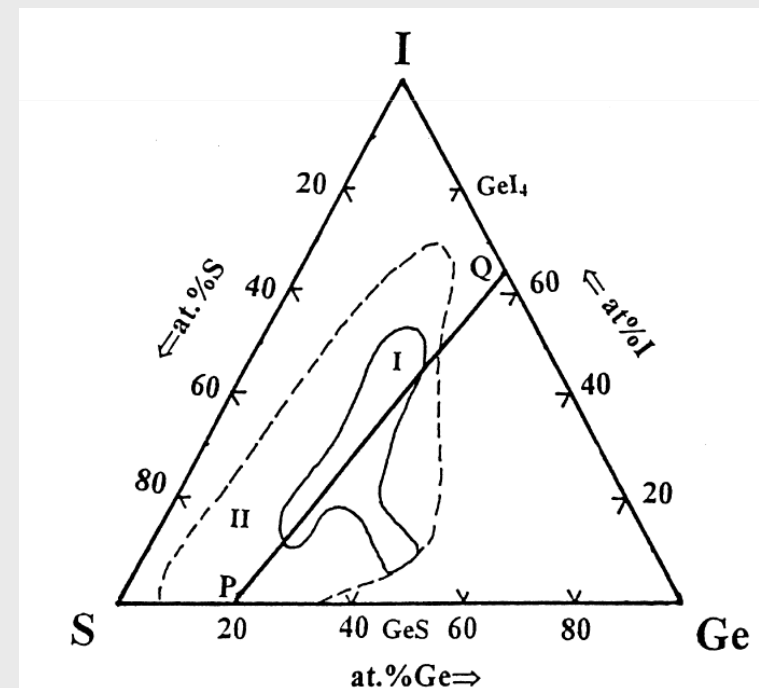
$$F/N = 6 - 5/2 \langle r \rangle - n_1/N$$

with $\langle r \rangle = \sum_{r \geq 1} r n_r / N$ (mean coordination number)

\Rightarrow structure is rigid if $F=0$

$\Rightarrow \langle r \rangle = 2.4 - 0.4 n_1/N$

\Rightarrow on this line glasses form easily



The mode-coupling theory of the glass transition (MCT)

see talks of Bagchi, Miyazaki, and Bhattacharyya

- Consider a **system which has degrees of freedom that are fast and slow** (good separation of time scales); the **Zwanzig-Mori projection operator formalism** (1960, 1965) is a method to derive **exact equations of motions for the slow dof** (by eliminating the fast dof's)
- Glasses: Vibrations (inside the cages) are fast; α -relaxation is slow
 \Rightarrow MZ formalism + approximations gives MCT equations

Typical structure of MZ equation: $\phi(q,t)$ = intermediate scattering function for wave-vector q

$$\ddot{\phi}(q, t) + \Omega^2(q)\phi(q, t) + \Omega^2(q) \int_0^t M(q, t-s)\dot{\phi}(q, s)ds = 0 \quad \text{with} \quad \Omega^2(q) = \frac{q^2 k_B T}{mS(q)}$$

This equation is exact but $M(q,t)$ is horribly complicated \Rightarrow make MCT approximations

$$M^{MC}(q, t) = \int d^3 q' V(q, q') \phi(q', t) \phi(q, t)$$

The mode-coupling theory: 2

$$\ddot{\phi}(q, t) + \Omega^2(q)\phi(q, t) + \Omega^2(q) \int_0^t M(q, t-s)\dot{\phi}(q, s)ds = 0 \quad \text{with} \quad \Omega^2(q) = \frac{q^2 k_B T}{mS(q)}$$

with
$$M^{MC}(q, t) = \int d^3q' V(q, q')\phi(q', t)\phi(q, t)$$

N.B.:

1: By the Z-M construction, the vertices $V(q, q')$ depend only on static quantities, such as the density, structure factor, three point correlation functions, ...

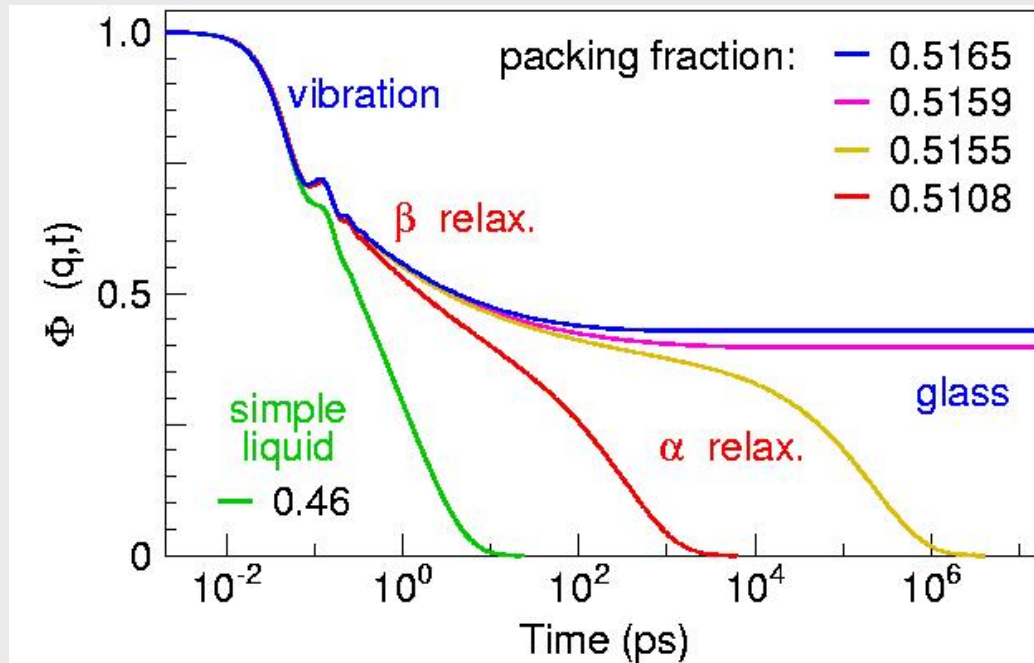
⇒ THE STATICS GIVES THE DYNAMICS!

2: If $S(q)$ becomes more peaked, $V(q, q')$ increases, i.e. the memory function increases with increasing density or decreasing temperature.

⇒ With increasing coupling the dynamics is slowed down and ultimately the system can arrest completely ⇒ **glass transition**

Mode-coupling theory: 3

- Consider the MCT solution for a very simple system: **hard spheres**



- qualitatively the curves resemble the ones found in experiments

- There exists a **critical temperature T_c** (or packing fraction) at which the relaxation times increase very quickly
- Close to T_c the **relaxation times show a power-law dependence:**

$$\tau \propto (T - T_c)^{-\gamma}$$

Mode-coupling theory: 4

- The MCT equations **are not exact for structural glasses**
- In 1986 Kirkpatrick, Thirumalai, and Wolynes studied certain **mean-field spin glass models** (see talks by Dasgupta, Mezard, and Biroli)

$$H = -\frac{1}{2} \sum_{i \neq j}^N J_{ij} (p \delta_{\sigma_i \sigma_j} - 1) \quad \text{with } \sigma_i \in \{1, \dots, p\}$$

They were able to derive **exact equations of motion for $C(t)$** , the spin-autocorrelation function: $C(t) = \langle \sigma_i(t) \sigma_i(0) \rangle$

These equations have the same mathematical structure as the MCT equations!

Conclusions:

- 1. There exist models for which the MCT equations are exact**
- 2. There might be a close connection between spin glasses and structural glasses**

Remark: Cugliandolo and Kurchan generalized 1993 these schematic models to the out-of-equilibrium case (fluct. dissipation theorem is no longer valid!) \Rightarrow **Theory for the dynamics of systems in the glass phase** (see talks by Kurchan and Franz)

Random First Order Theory

(see talks by Dasgupta, Mezard, Biroli, Bhattacharyya)

Kirkpatrick, Thirumalai, and Wolynes 1980's: Studied the **mean field p-spin model**:

$$H = - \sum_{i,j,k} J_{ijk} \sigma_i \sigma_j \sigma_k$$

⇒ **Landscape has at low T** (but above T_K) **many different valleys** (see also results from replica formalism) ⇒ configuration space can be decomposed into **metastable states** that have infinite lifetime

⇒ partition function can be written as sum over states:

$$Z = \sum_{\sigma} \exp(-\beta H(\sigma)) = \sum_{\alpha=1}^L \sum_{\sigma \in \alpha} \exp(-\beta H(\sigma)) = \sum_{\alpha=1}^L \exp(-\beta f_{\alpha})$$

$$= \int df \exp(-\beta N[f - TS_c(f)]) \quad \text{with} \quad S_c = N^{-1} \log \sum_{\alpha=1}^L \delta(f - f_{\alpha})$$

⇒ Saddle point approximation ⇒ extremum f^* with $S'_c(f^*)=1/T$

Minimum of free energy is $F = f^* - TS_c(f^*)$

Random First Order Theory: 2

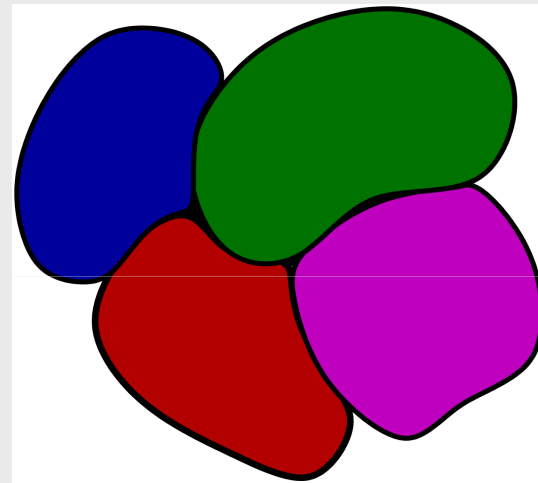
Finite dimensional systems (= finite range of interactions): The metastable states from MF still survive (but have now a finite lifetime)

⇒ different parts of the system will be in different states

⇒ **decomposition of the system into patches/tiles that correspond to a local minimum of the free energy**

⇒ There is an **interface between these patches** which costs energy

⇒ **surface tension $Y(T)$**



⇒ A patch of size R will have **free energy gain ΔF_{gain}** and an **interface cost ΔF_{cost}** with

$$\Delta F_{\text{gain}} = -TS_c(R)R^d$$

$$\Delta F_{\text{cost}} = Y(T)R^\theta \quad \text{with} \quad \theta \leq d - 1$$

Random First Order Theory: 3

The **size of the patches** ξ will be given by the condition

$$\Delta F_{\text{gain}} = \Delta F_{\text{cost}} \implies \xi = \left(\frac{Y(T)}{TS_c(T)} \right)^{\frac{1}{d-\theta}}$$

\implies length scale that grows with decreasing T!

Connection to dynamics:

Height of free energy barrier Δ to flip a state/patch:

$$\max \Delta F = \max Y(T)R^\theta - TS_c R^d \implies \Delta = \frac{Y(T)^{\frac{d}{d-\theta}}}{[TS_c(T)]^{\frac{\theta}{d-\theta}}}$$

Assume that process is activated \implies **relaxation time** τ is given by

$$\tau = \tau_0 \exp(\Delta/T) = \tau_0 \exp \left(\frac{Y(T)^{\frac{d}{d-\theta}}}{T[TS_c(T)]^{\frac{\theta}{d-\theta}}} \right)$$

If we chose for the **exponent** θ

we get the expression by Adam and Gibbs

$$\theta = d/2 \implies \tau = \tau_0 \exp \left(\frac{C}{TS_c(T)} \right)$$

Summary

!!! CAUTION !!! DON'T GET FOOLED !!!

- There are **many approaches** that attempt to describe the structure and the glassy dynamics: Some of them are highly sophisticated, some of them are simple minded
- **All of the non-trivial approaches have flaws:**
 - **Fuzzy concepts:** What are the cooperatively rearranging regions of Adam-Gibbs? Does it make sense to talk about an interface tension in the RFOT if the domains are small?, ...
 - **Uncontrolled approximations:** MCT takes hopping processes into account in a rudimentary way. What is the relevance of mean field results for finite dimensional systems
- **Theories have helped us to make significant progress in our understanding of glass-forming systems, but still there is a lot to do**