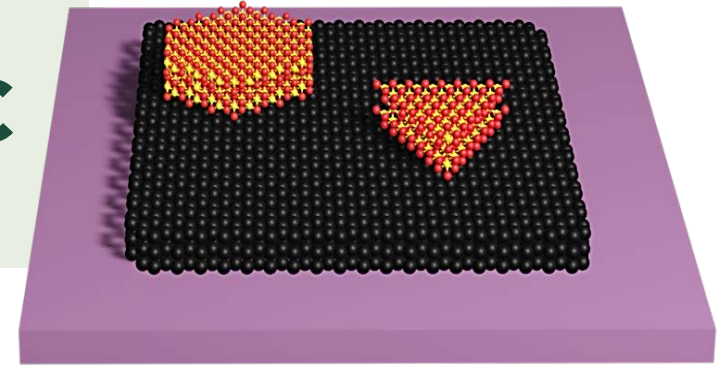
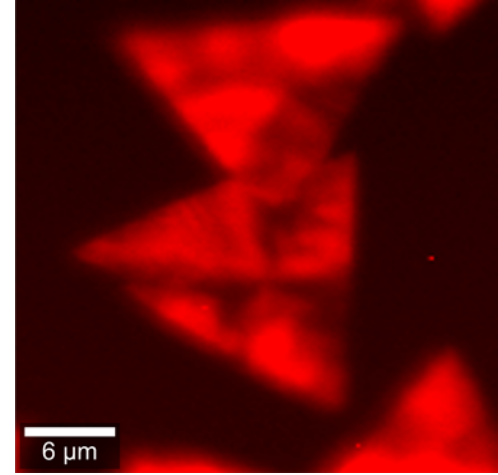
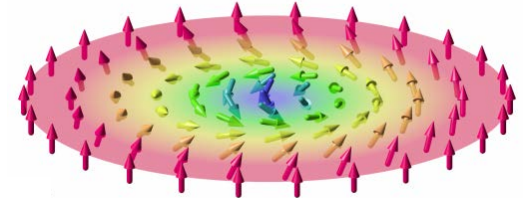
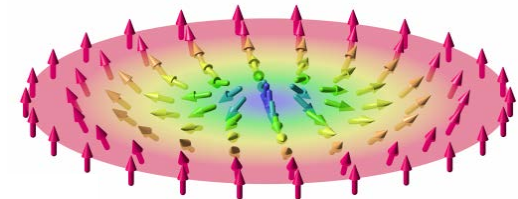


# VAN DER WAALS EPITAXIAL GROWTH OF 2D/QUASI-2D MATERIALS & THEIR PROSPECTS FOR OPTOELECTRONIC AND SPINTRONIC DEVICES



Vidya Kochat  
IIT Kharagpur  
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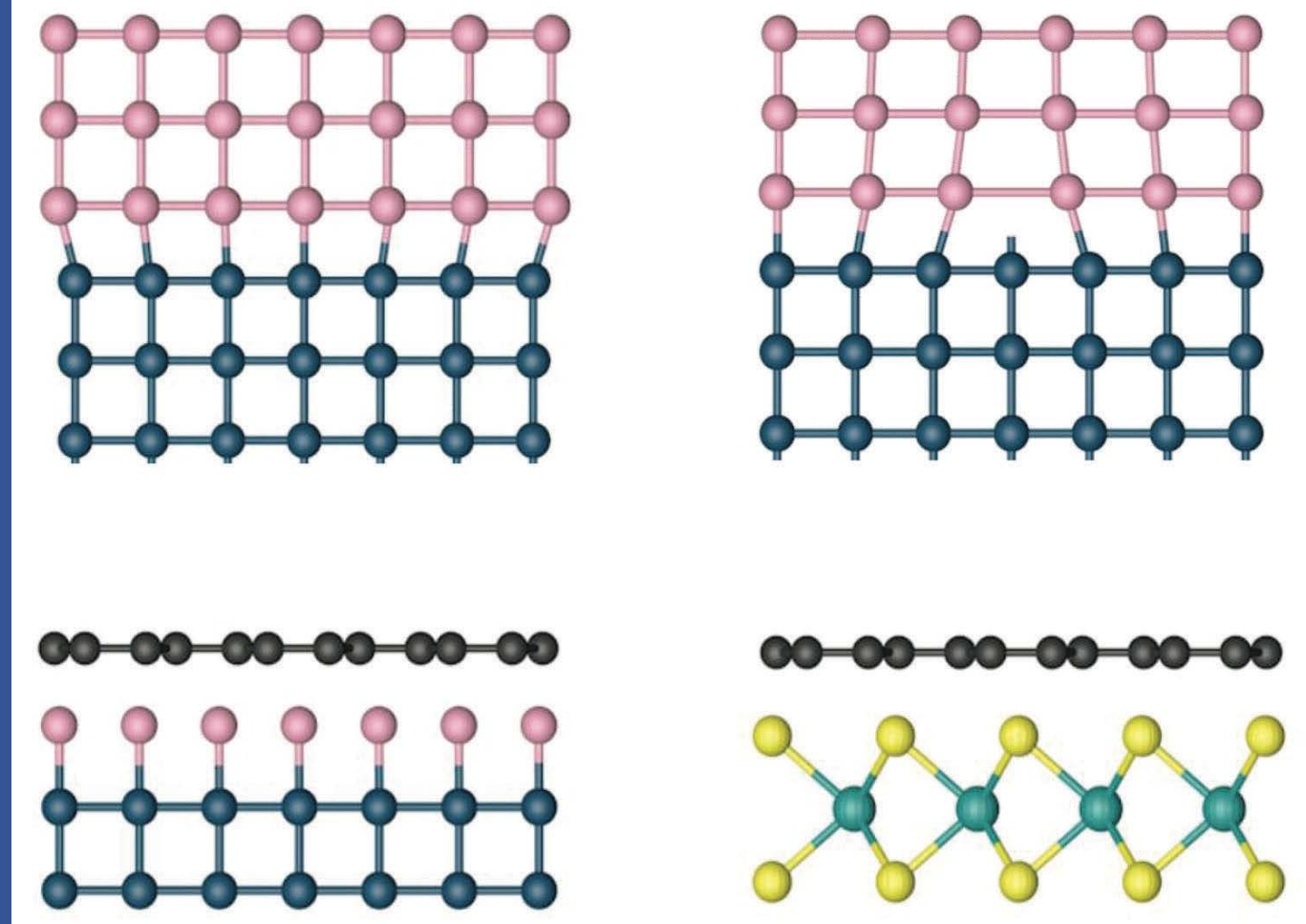


# van der Waals epitaxy

vdW epitaxy is the growth mechanism of epitaxial layers on crystalline substrates governed by weak van der Waals forces between the epilayer and substrate.

Heterointerfaces with negligible strain, despite large lattice mismatch and thermal expansion coefficients.

## Conventional epitaxy



## van der Waals epitaxy



# Talk outline

## 01

Epitaxy of 2D/  
quasi-2D materials  
growth

## 02

Quasi-2D magnet -  $\text{Cr}_{1+x}\text{Te}_2$   
Growth, non-collinear  
magnetism and prospects  
for spintronics

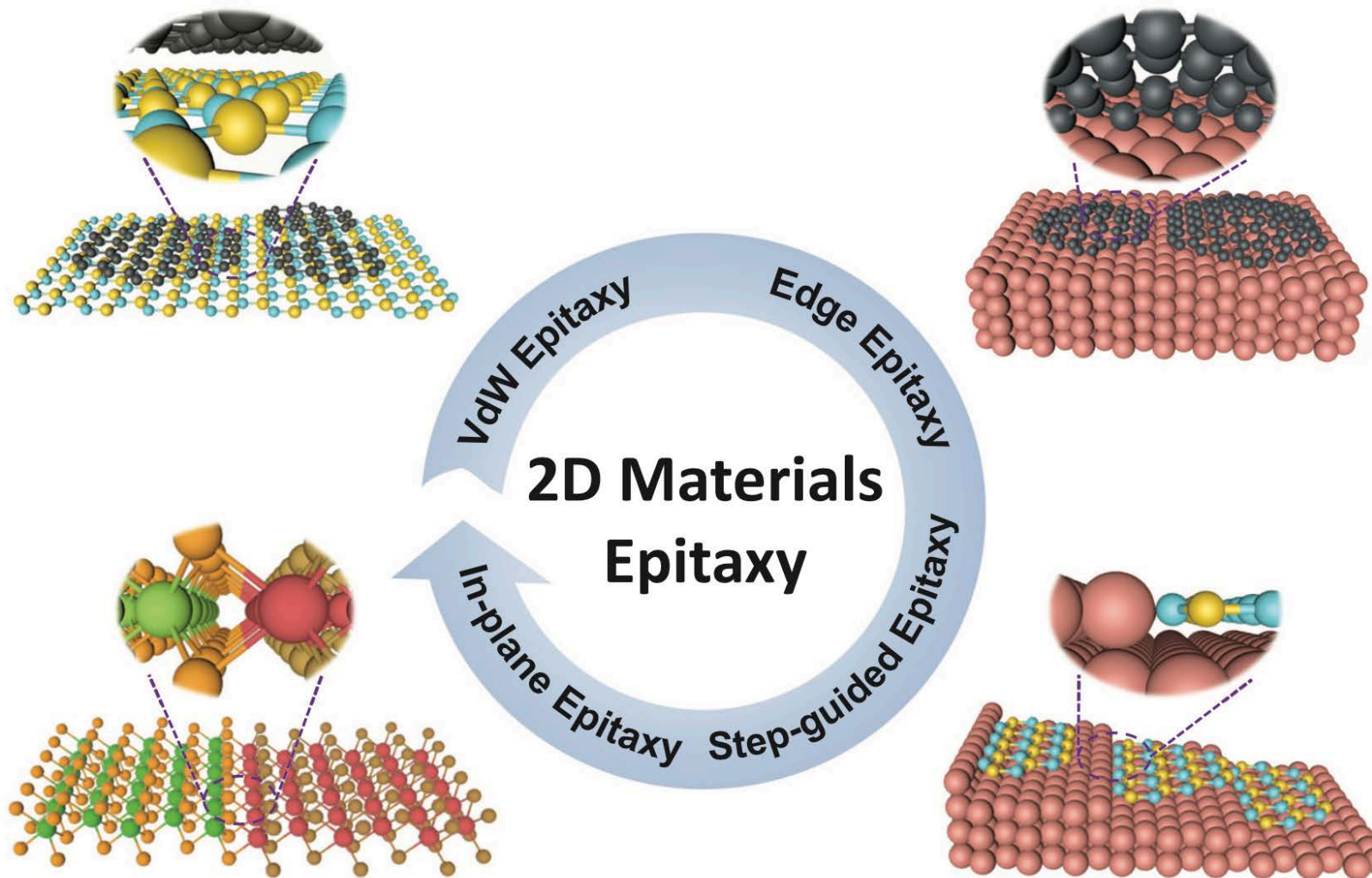
## 03

2D semiconductors -  $\text{WSe}_2$   
Optical emission  
signatures and prospects  
for optoelectronics

## 04

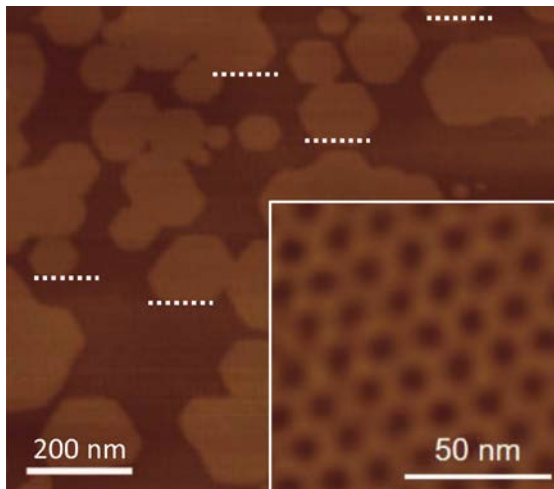
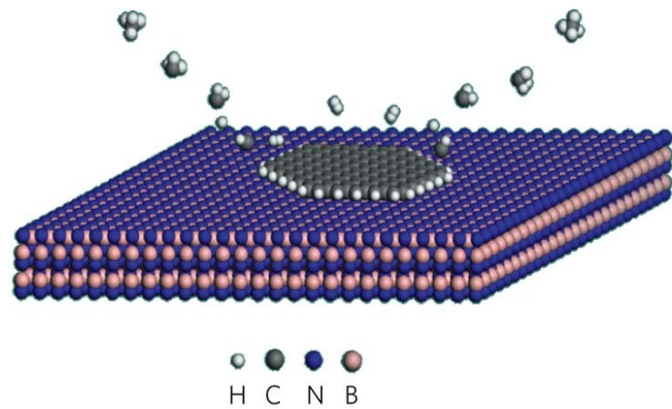
Conclusions  
The future

# Epitaxy of 2D materials growth



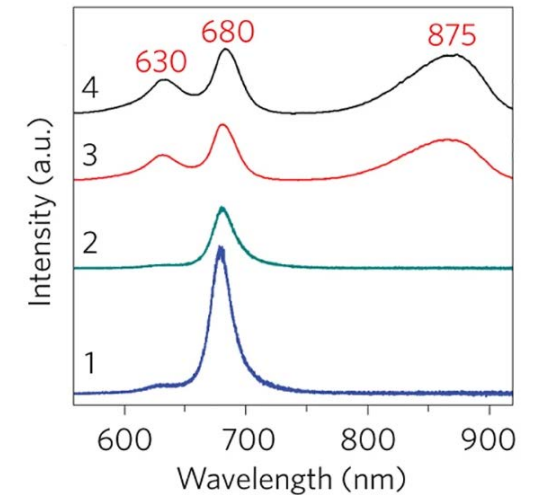
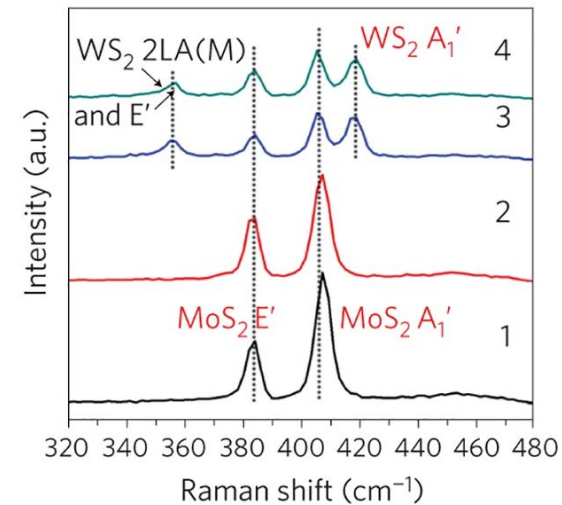
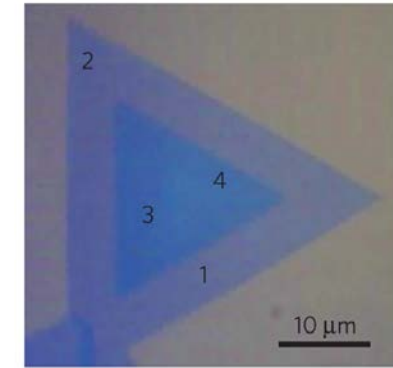
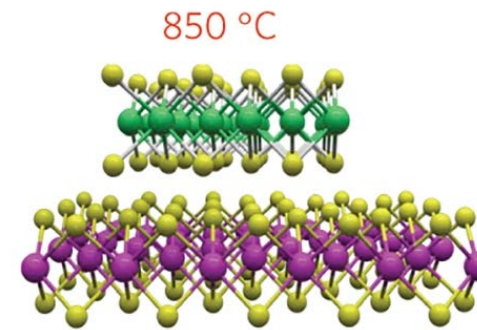
# van der Waals epitaxy

Epitaxial growth of graphene on hexagonal boron nitride



*Nat. Mater.* **2013**, 12, 792

Vertical heterostructures from WS<sub>2</sub>/MoS<sub>2</sub> monolayers

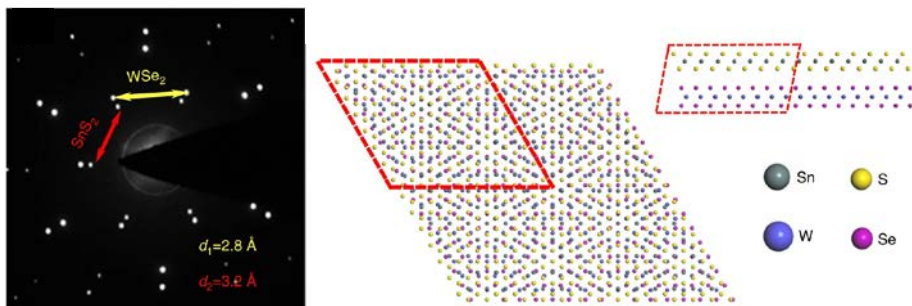
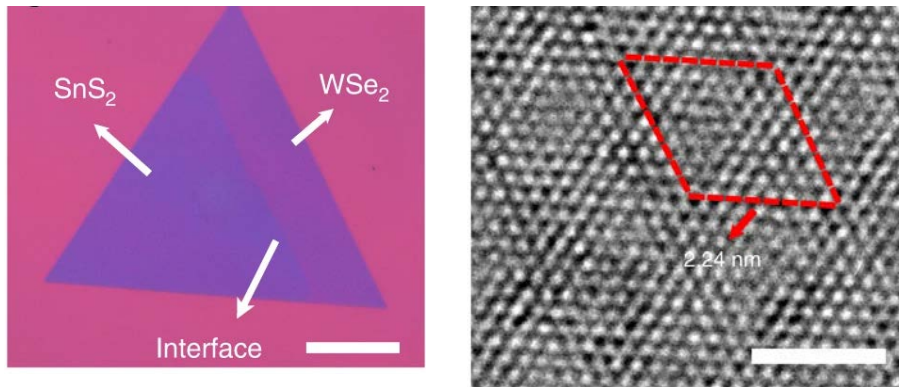


*Nat. Mater.* **2014**, 13, 1135

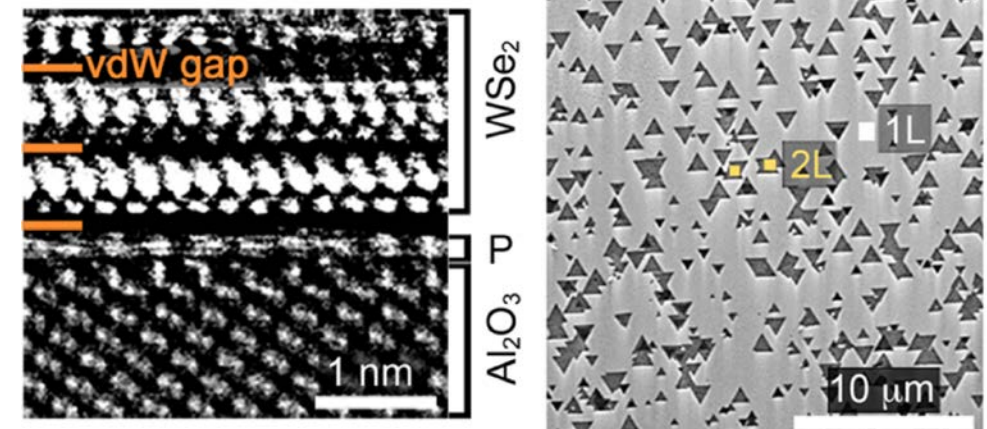
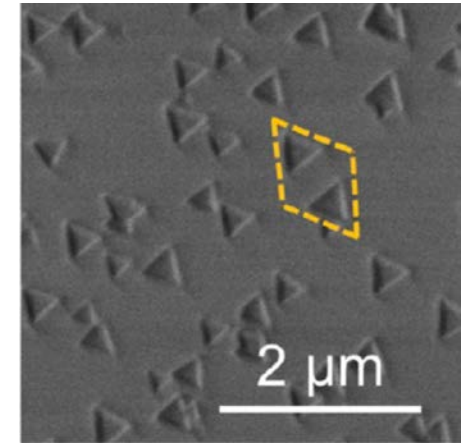
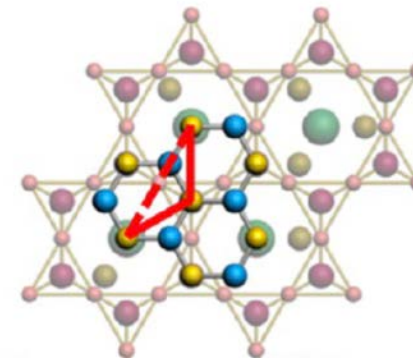
# van der Waals epitaxy

Possible with large lattice mismatch (as high as 50%) and between different crystal structures

## WSe<sub>2</sub>/SnS<sub>2</sub> vertical heterostructures



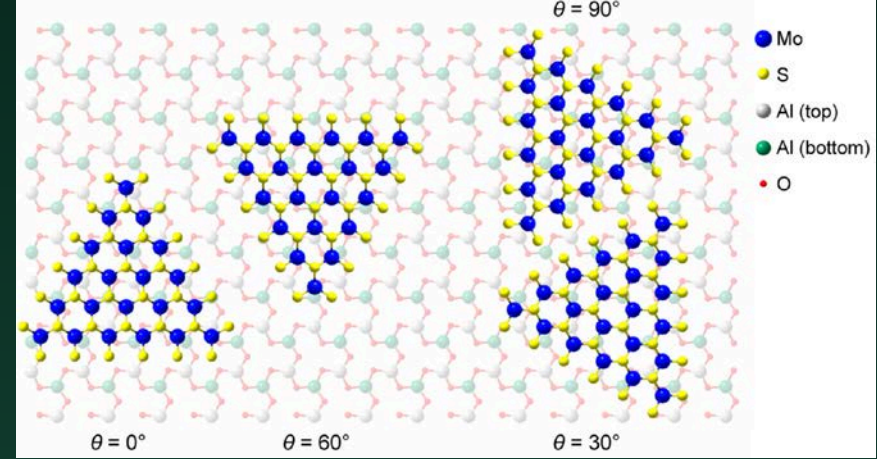
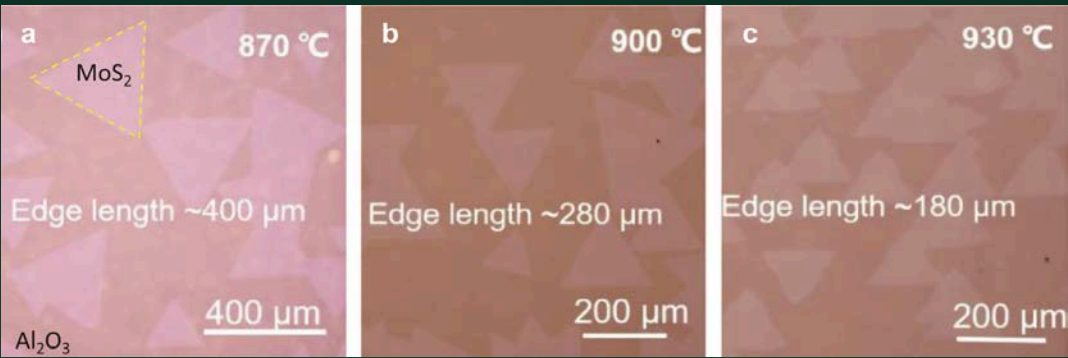
## Epitaxial TMD monolayers on mica and sapphire



Nat. Commun. **2017**, 8, 1906

Nano Lett. **2013**, 13, 3870  
ACS Nano **2018**, 12, 965

# Modulating vdW epitaxy



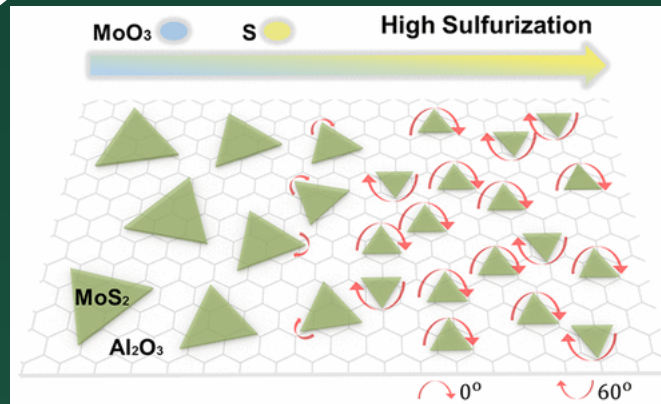
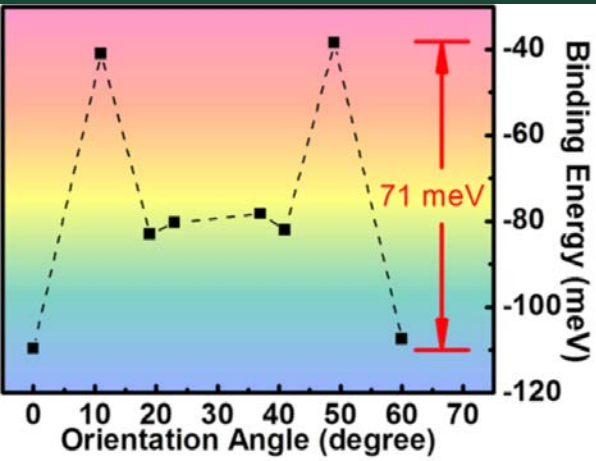
ACS Nano 2015, 9, 4611

Nano Lett. **2020**, 20, 7193  
 Nano Lett. **2015**, 15, 198

Temperature

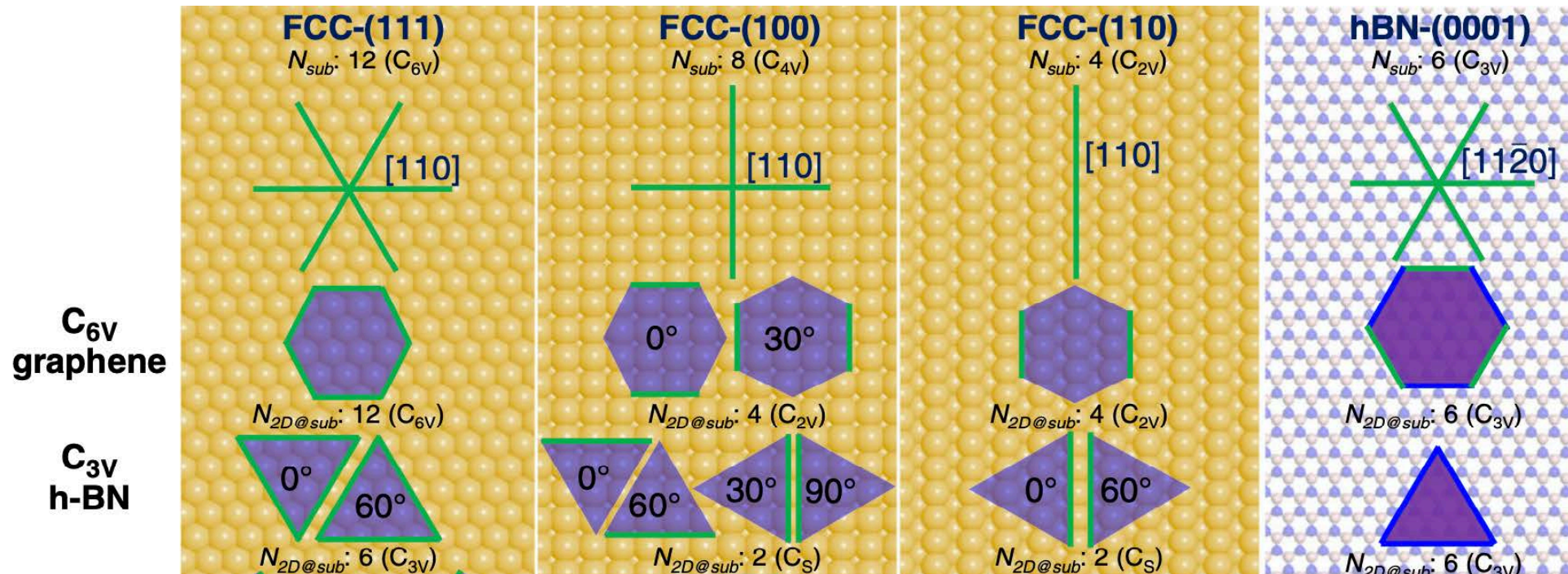
Precursor concentration

Substrate symmetry



ACS Nano **2017**, 11, 9215

# Interplay of the symmetries of the 2D material and the substrate



The propagating edge of a 2D material tends to align along a high symmetry direction of the substrate.

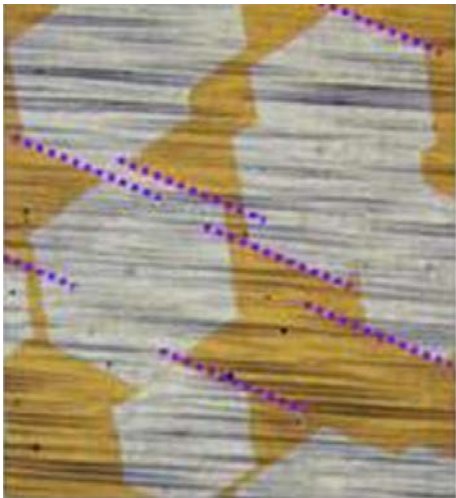
1, 2, 1 and 1 equivalent but different alignments for a 6-fold symmetric 2D material on 6-, 4-, 2- and 3-fold symmetric substrates  
 2, 4, 2, and 1 equivalent but different alignments for a 3-fold symmetric 2D material on 6-, 4-, 2- and 3-fold symmetric substrates



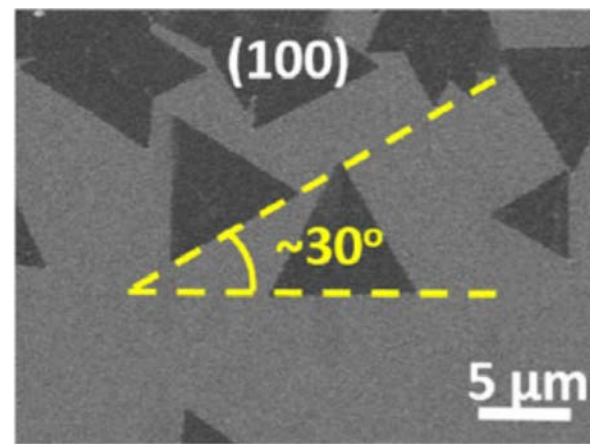
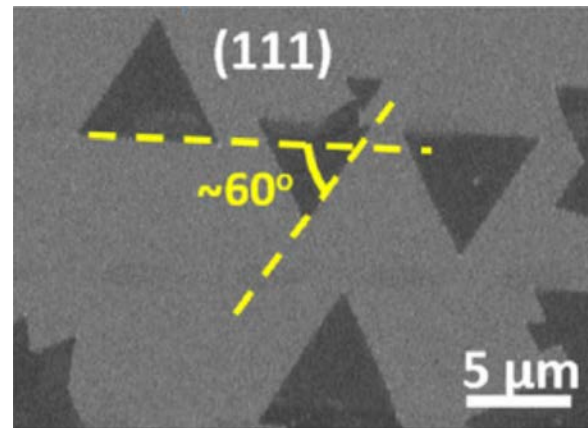
# Interplay of the symmetries of the 2D material and the substrate

## Experimental observations

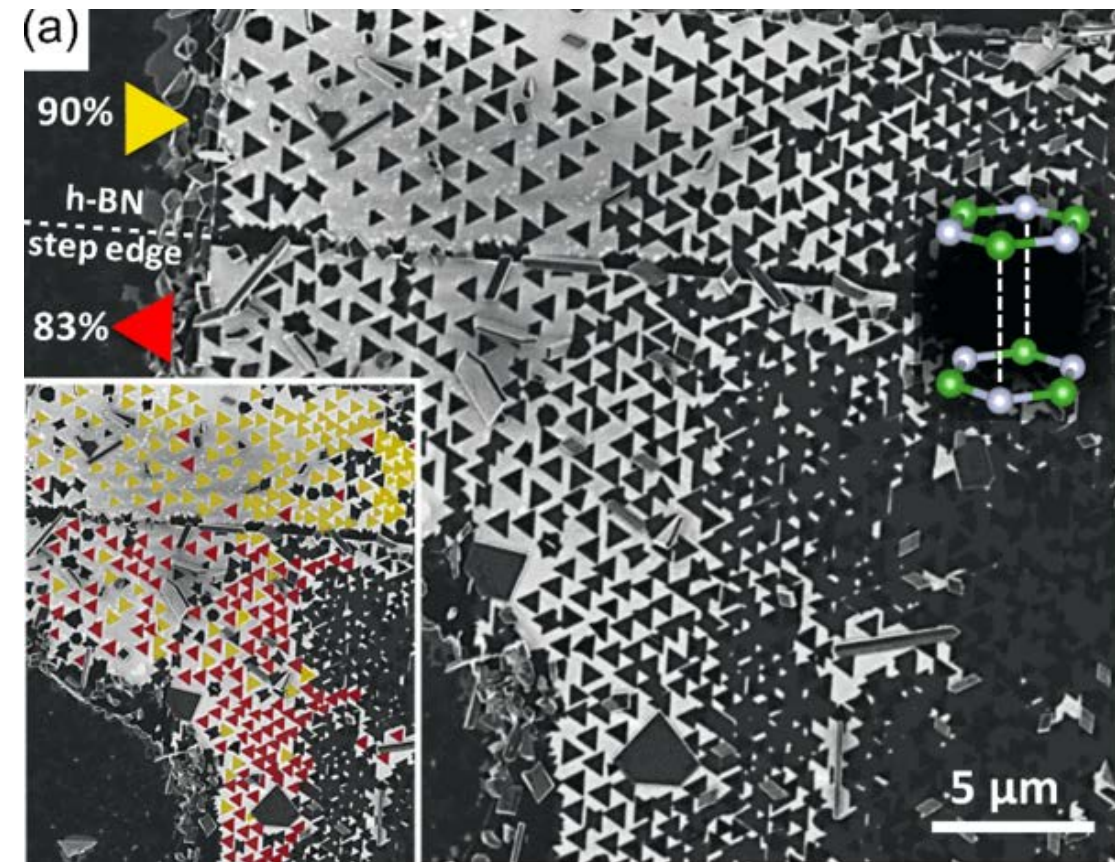
Graphene on Cu(111)



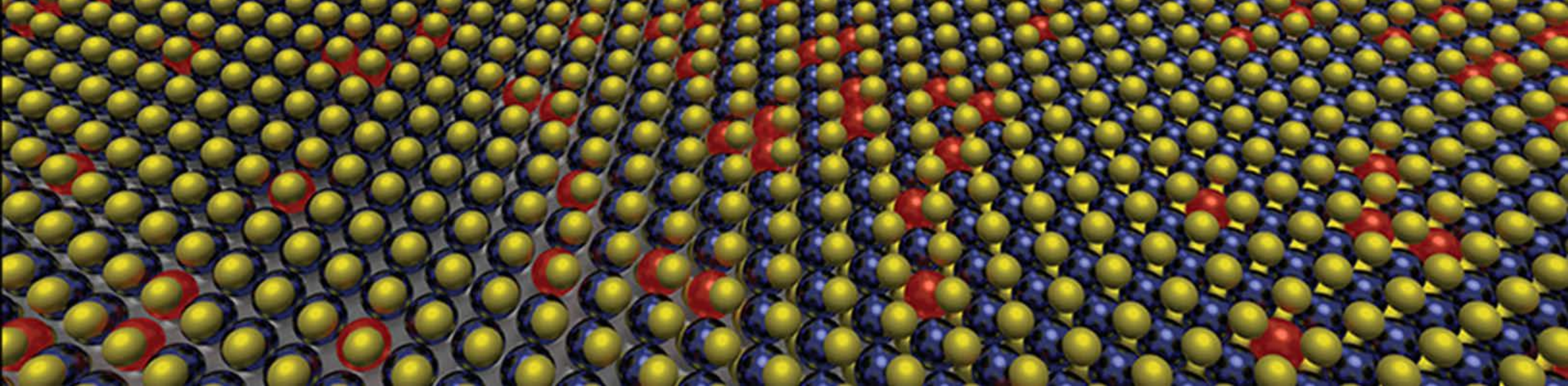
h-BN on Cu (111) & Cu (100)



MoS<sub>2</sub> on h-BN



*Sci. Bull.* **2017**, 62, 1074  
*Nano Res.* **2015**, 8, 3164  
*Phys. Rev. B* **2019**, 99, 155430

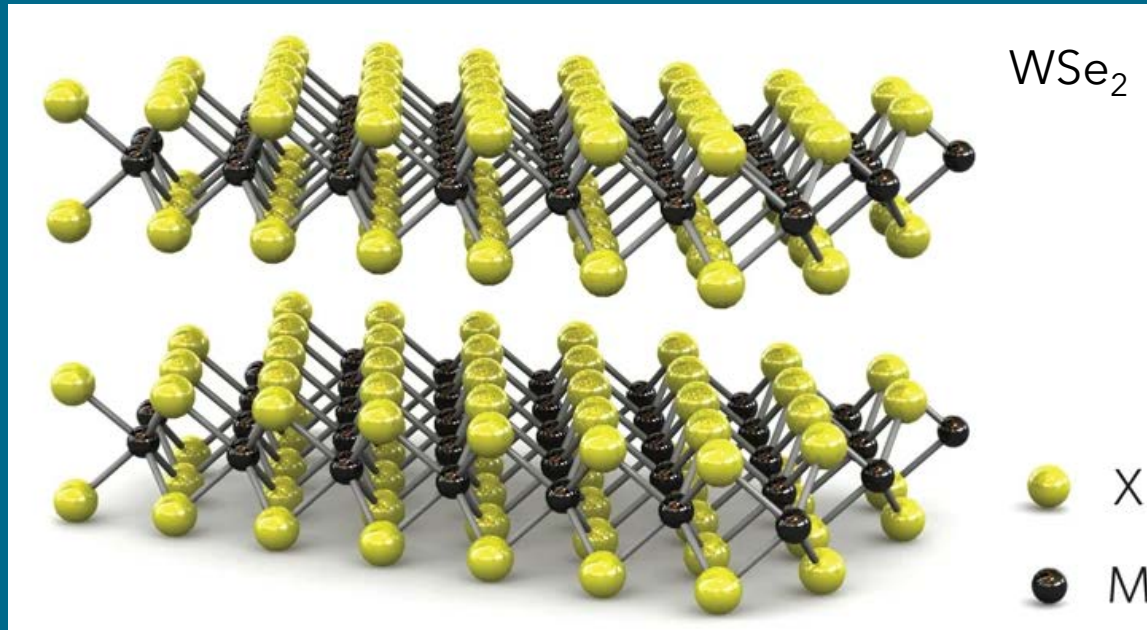


## Quasi-2D magnet - $\text{Cr}_{1+x}\text{Te}_2$

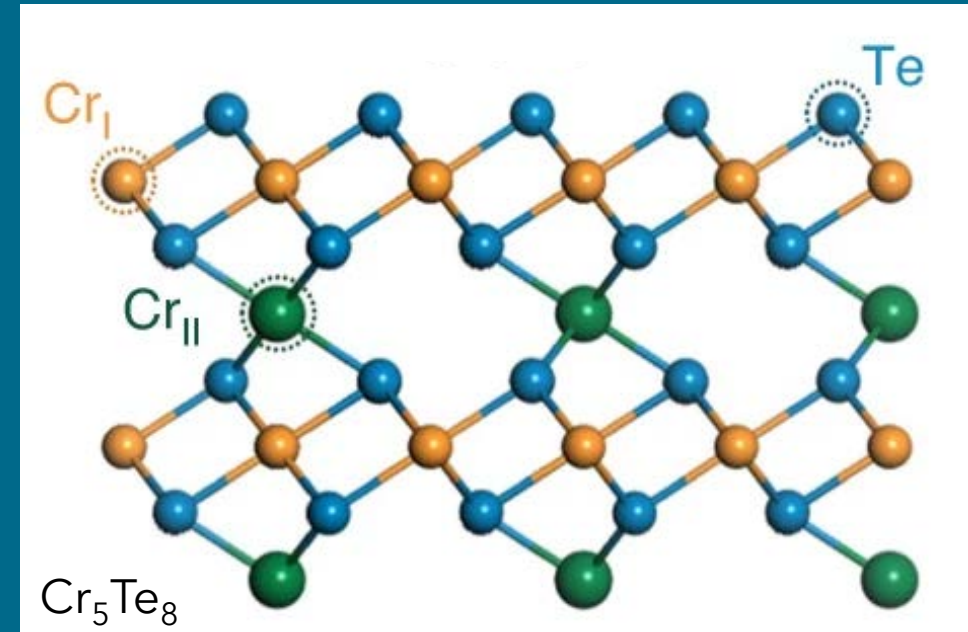
Growth, non-collinear magnetism and prospects for spintronics

# Quasi-2D materials

*Two dimensional materials manifest two distinct forms: layered and non-layered, based on the strength of the interlayer coupling effect.*



2D materials : atomically thin layers interconnected by vdW forces



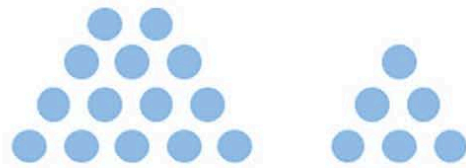
Quasi-2D materials : 2D non-vdW materials with covalently bonded intercalants between layers of unit cell thickness

# Epitaxial growth of quasi-2D materials

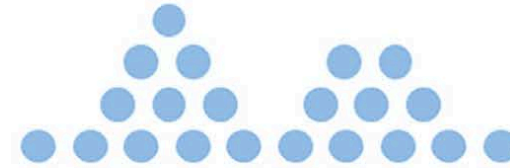
Frank-van der Merwe



Volmer-Weber

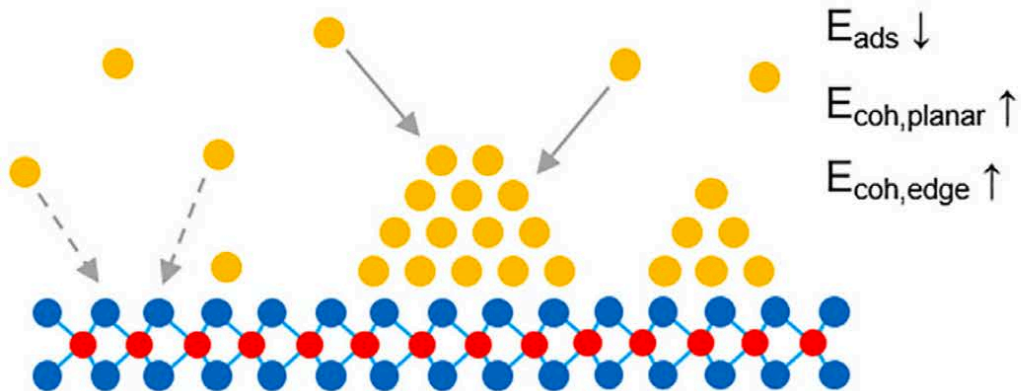


Stranski-Krastanov

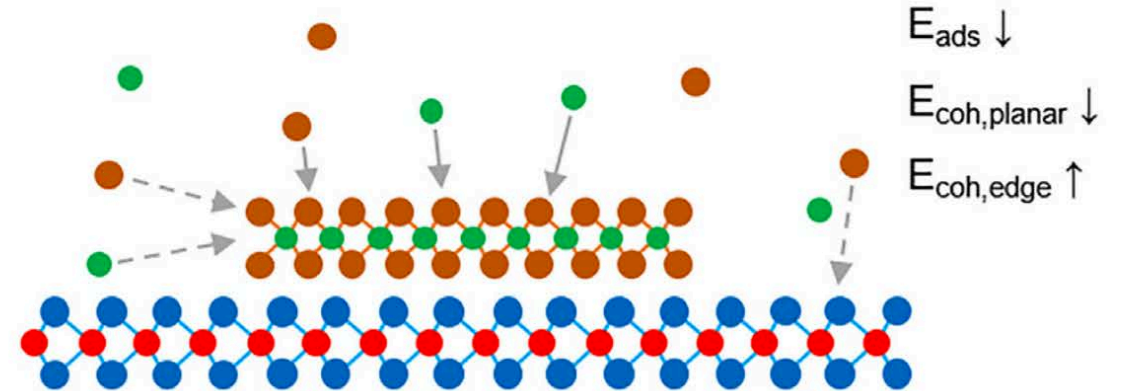


Substrate

(Volmer-Weber, Stranski-Krastanov)



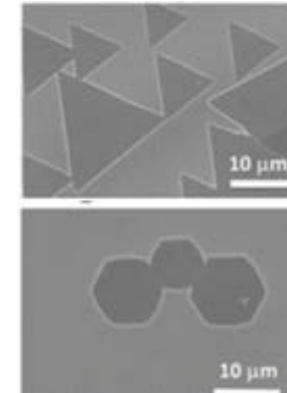
(Frank-van der Merwe)



# Kinetics of growth

## ATTACHMENT-LIMITED GROWTH

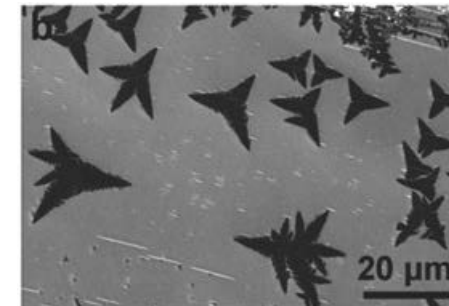
The precursor concentration on the whole substrate is nearly a constant and therefore the growth is dominated by the attachment of precursor atoms at the growing edges of a 2D material.



Single crystalline polygonal domains

## DIFFUSION-LIMITED GROWTH

In barrierless attachment of precursor to the edges of a 2D material, there will be a depletion zone around a growing 2D material and the growth must be determined by the diffusion of atoms/molecules on the substrate.



fractal-like shapes

$$R_G = \left( c \cdot \nu \cdot e^{-E_{ba}/kT} - n_0 \cdot \nu \cdot e^{-E_{bd}/kT} \right) \cdot s_0$$

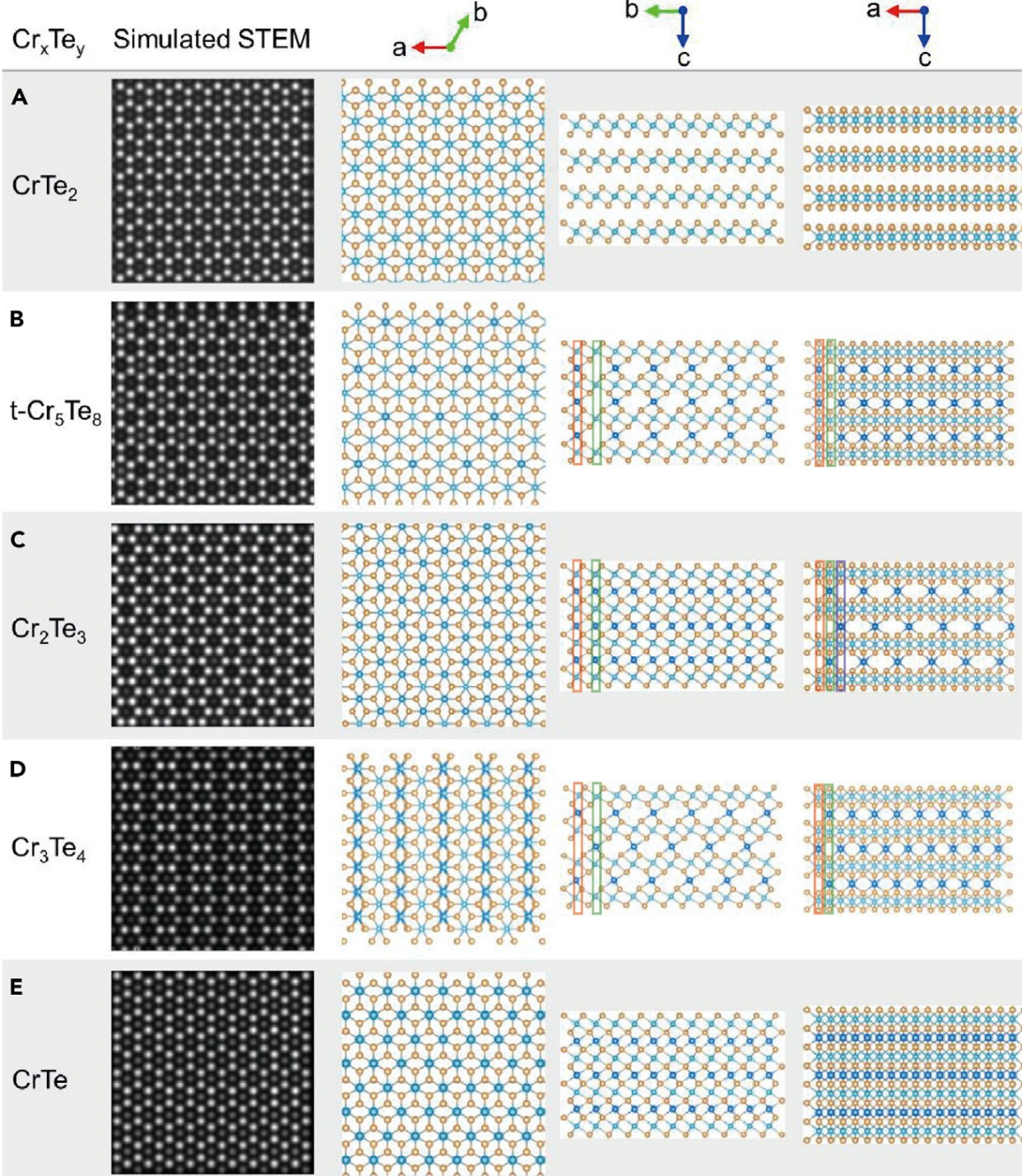
For growth of material :

$$E_{bd} - E_{ba} < kT \times \ln \left( \frac{c}{c_0} \right)$$

## Conditions for attachment-limited growth of 2D single crystals

- 1) Atomic attachment/detachment is very slow ( $E_{bd}$  &  $E_{ba}$  large)
- 2) Diffusion of atoms on the substrate is extremely fast
- 3) Atomic flux is sufficient large

# Quasi-2D magnet - $\text{Cr}_{1+x}\text{Te}_2$



$\bar{P}3m1$

1T  $\text{CrTe}_2$  - layered material

$\bar{P}3m1$

25%

$\text{Cr}_{1+x}\text{Te}_2$  compounds:  
Cr self-intercalation of the  $\text{CrTe}_2$   
backbones

$\bar{P}31c$

33.3%

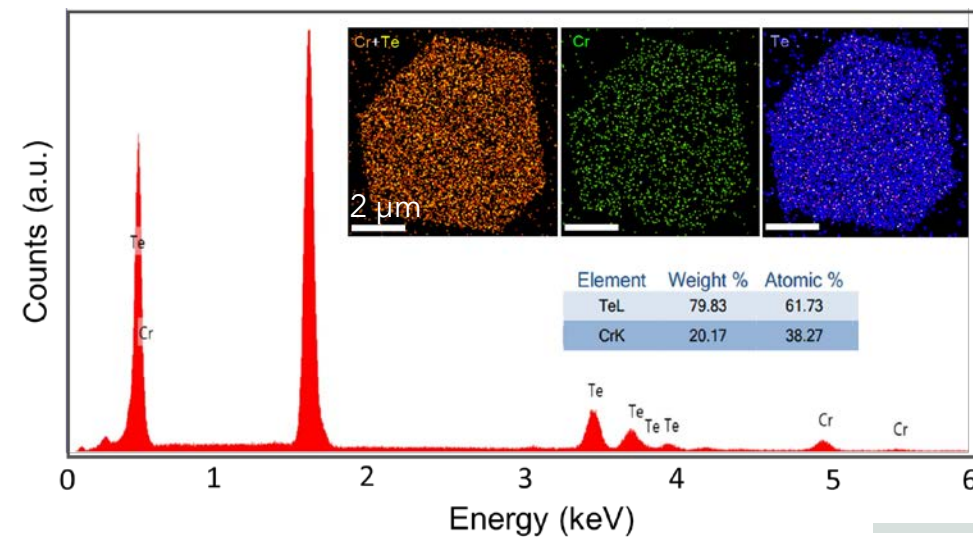
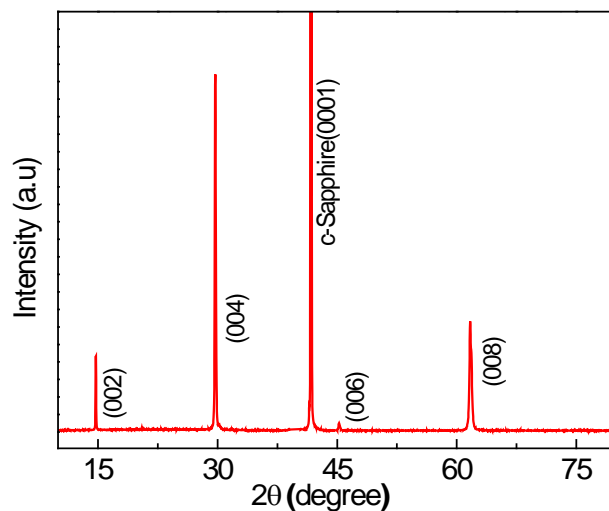
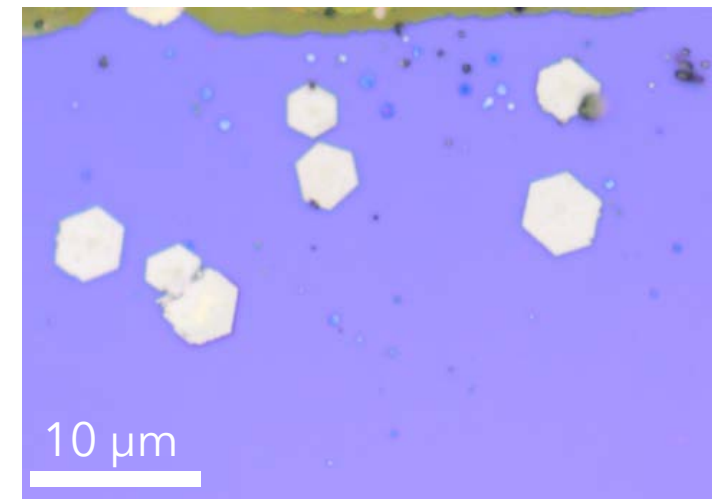
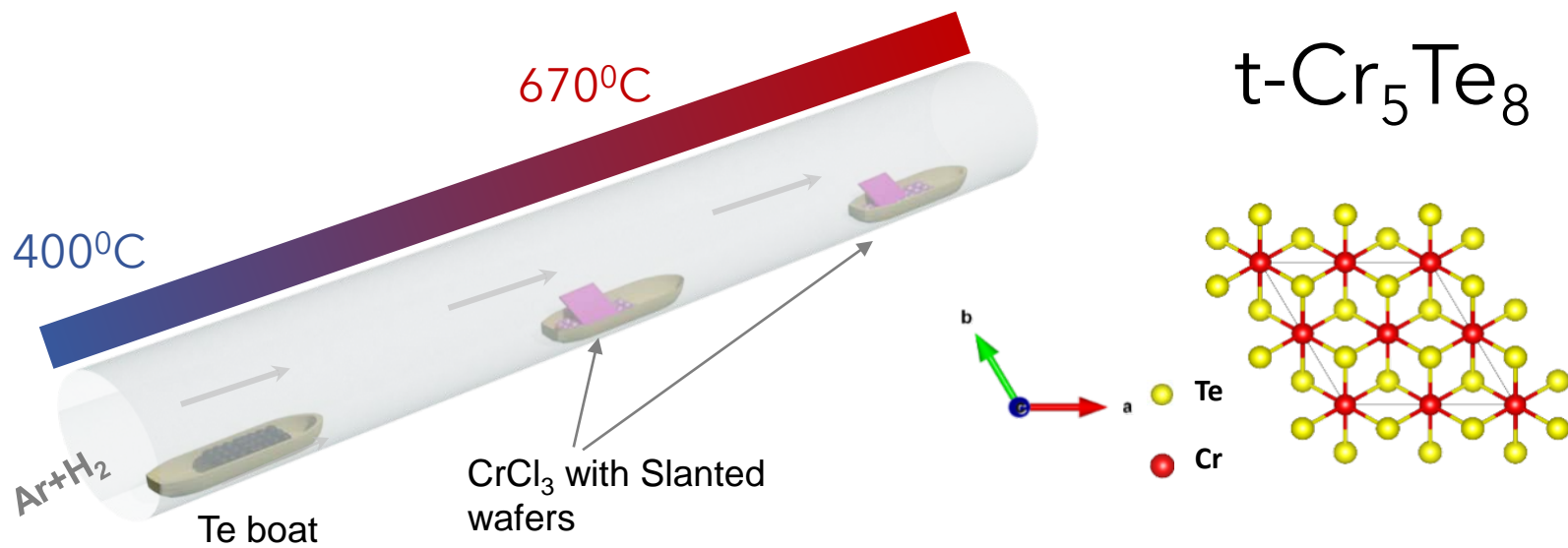
$P2/m$

50%

By self-intercalation of the native Cr atoms, the in-plane anisotropy of  $\text{CrTe}_2$  can be switched to perpendicular magnetic anisotropy.

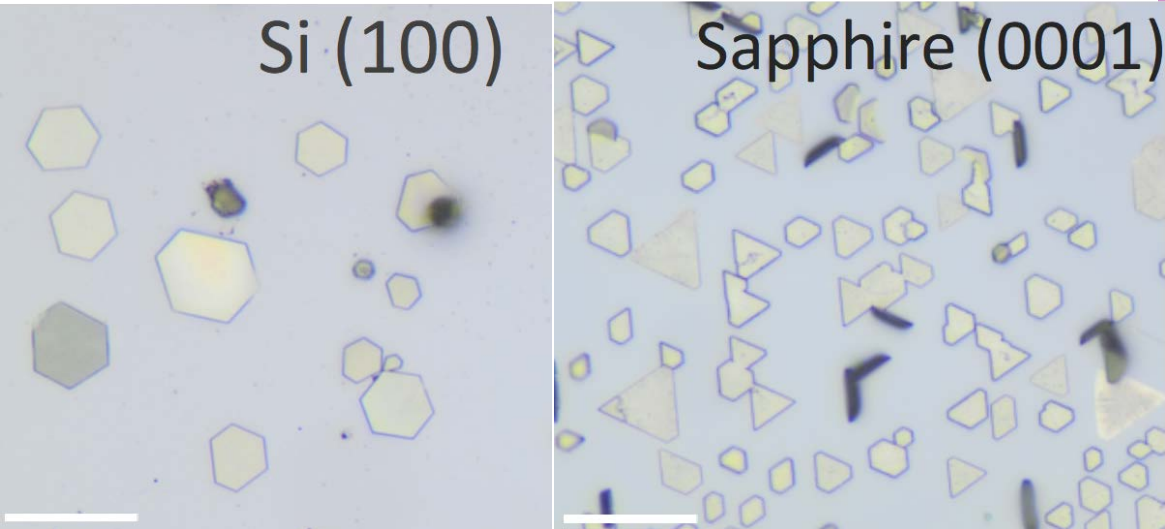
$P6_3/mmc$

# Growth of Quasi-2D $\text{Cr}_5\text{Te}_8$

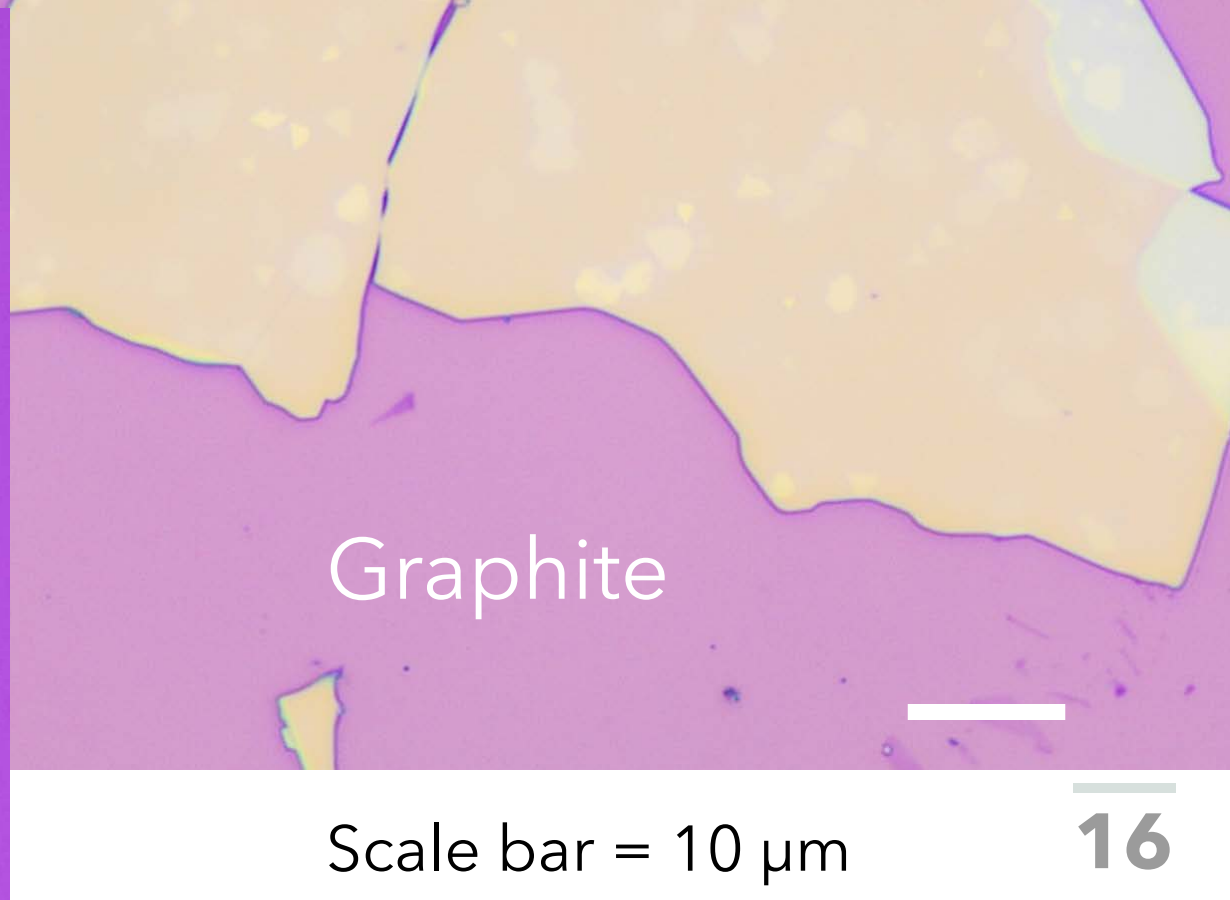
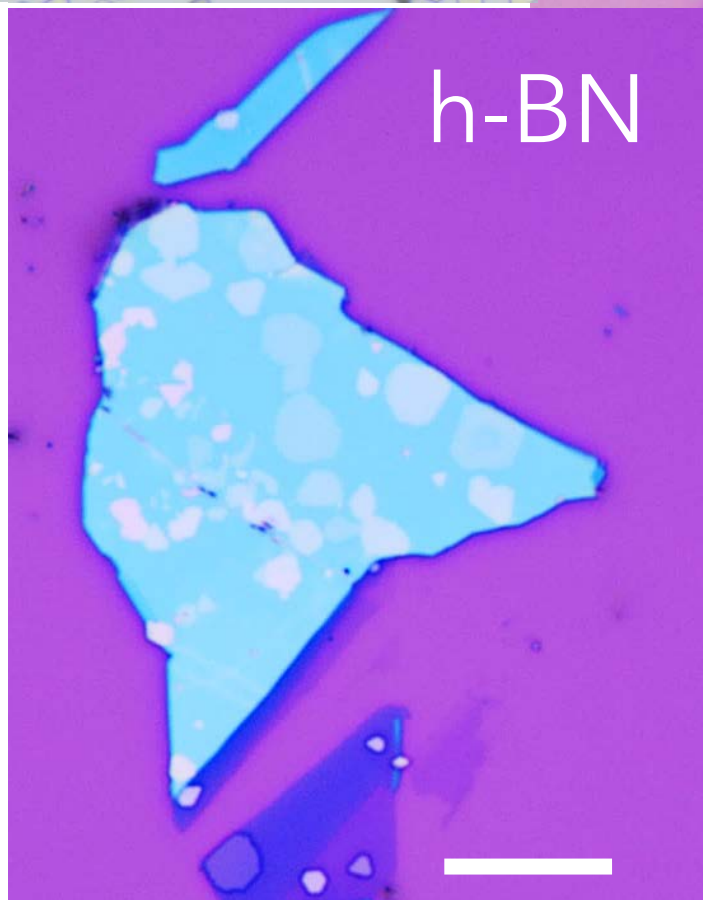
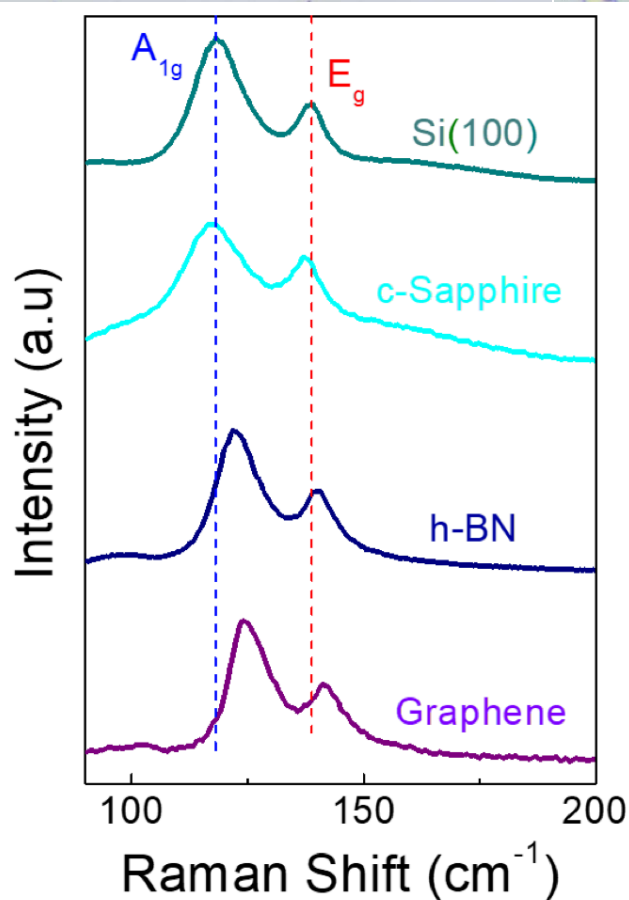


## Growth conditions

Precursors: 60 mg Te, 5 mg CrCl<sub>3</sub>  
Gas flow: 100 sccm Ar + 10 sccm H<sub>2</sub>  
Growth time : 4 min

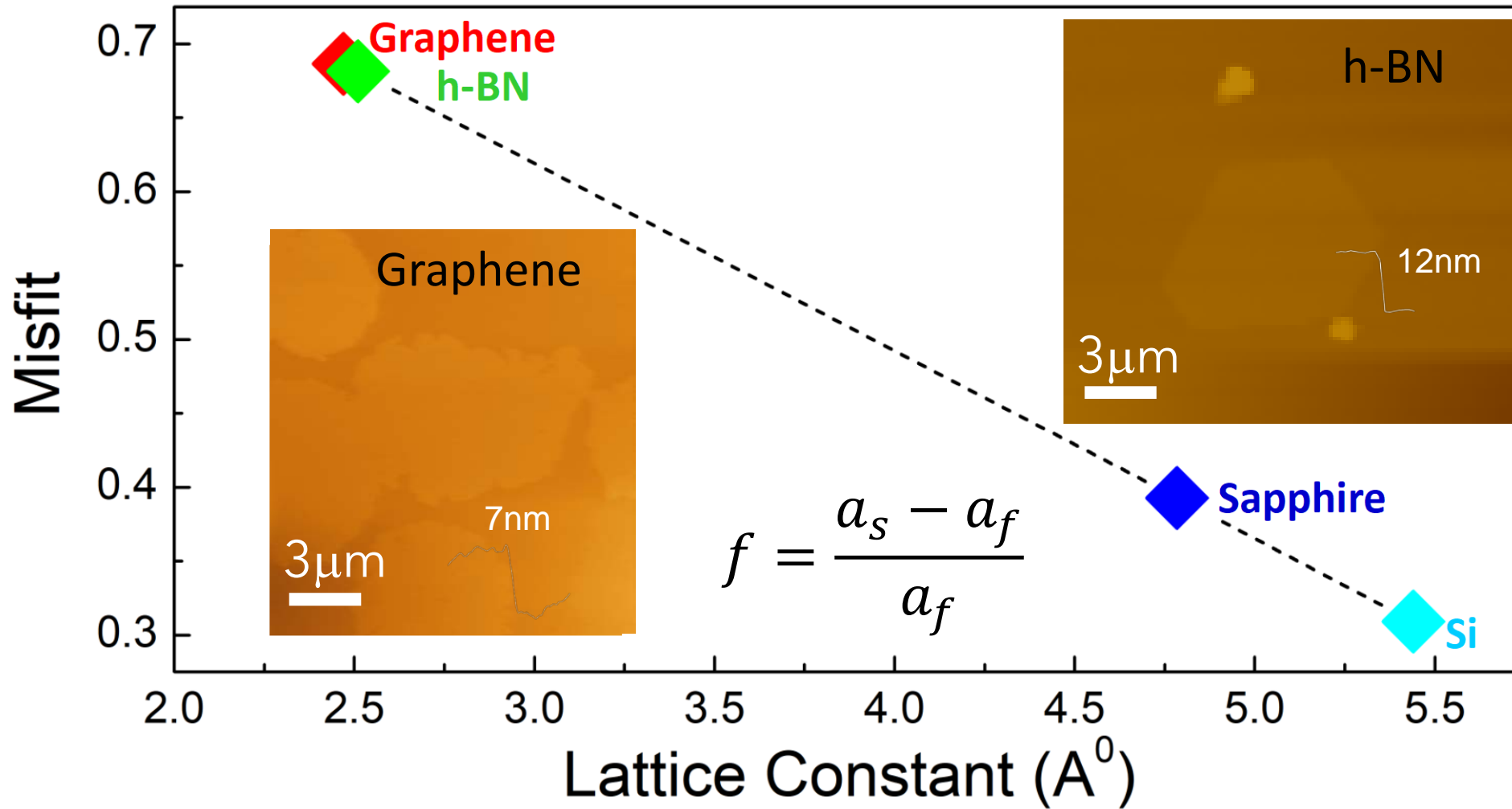


# Epitaxial growth on vdW substrates



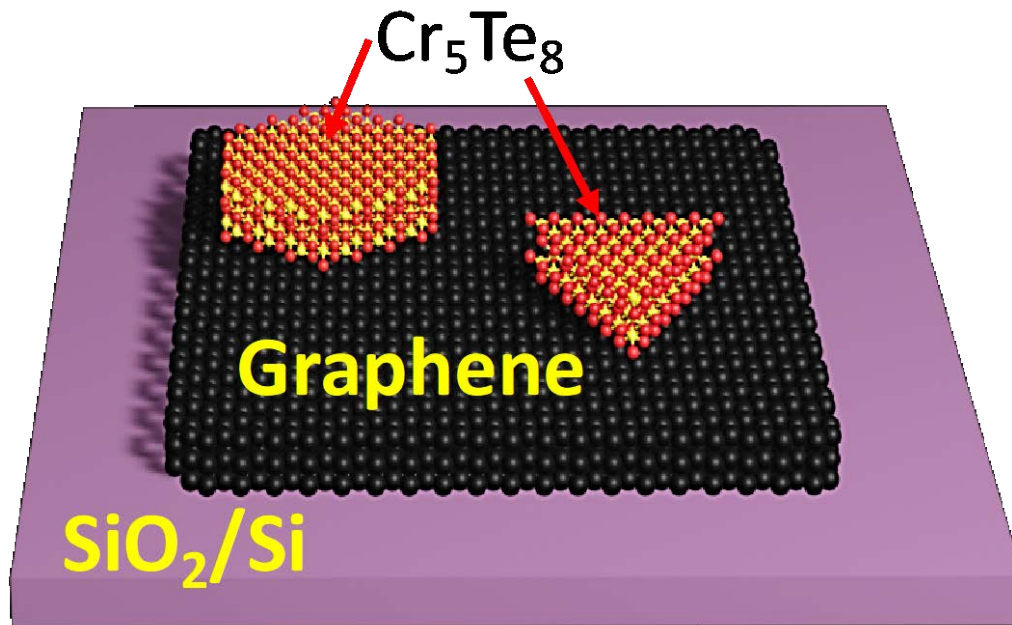


# Substrate criticality in vdW epitaxy

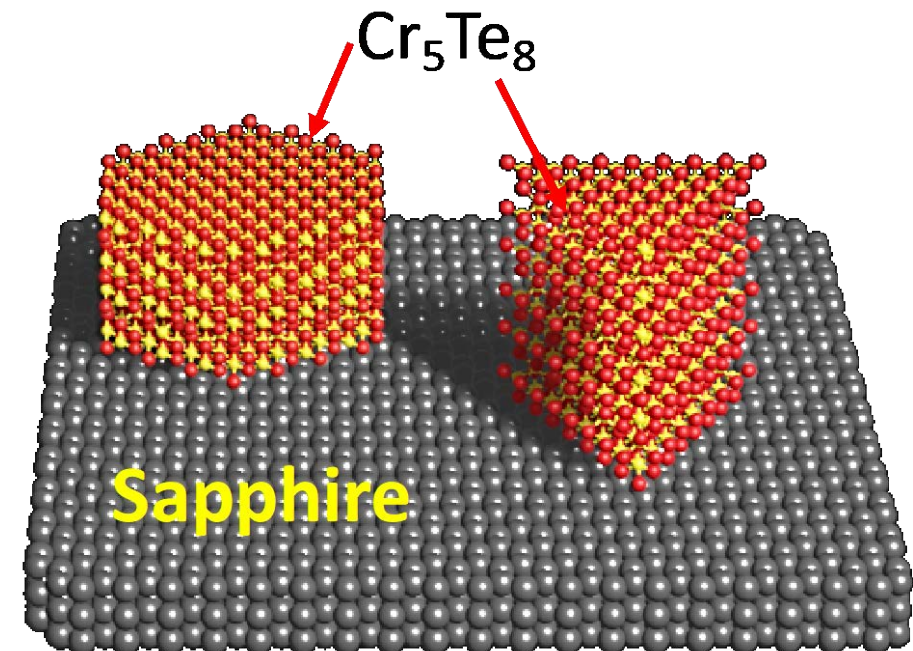


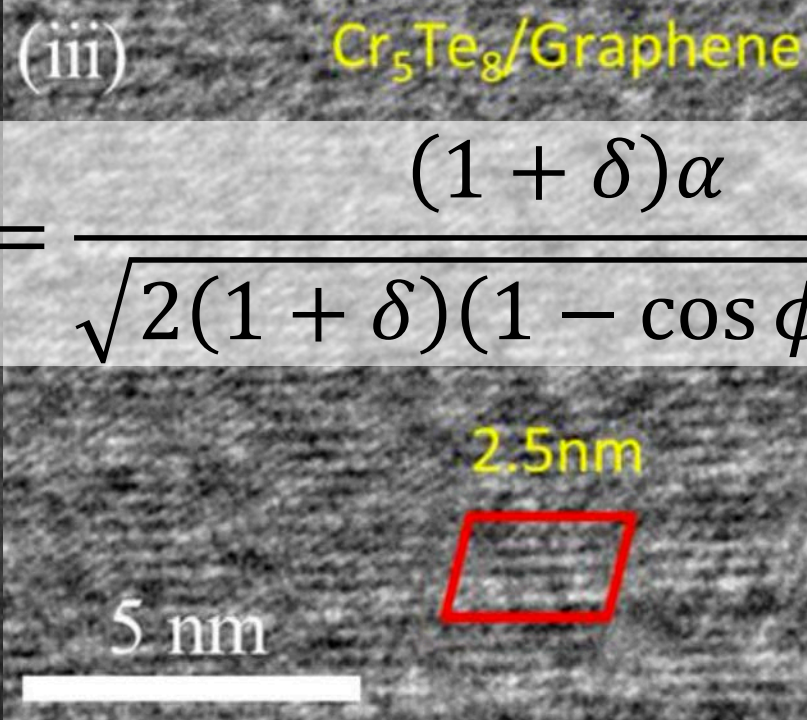
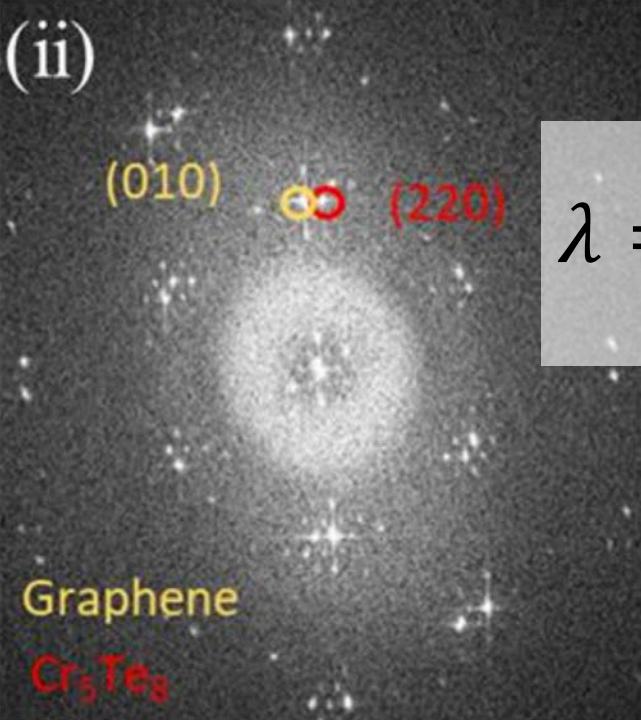
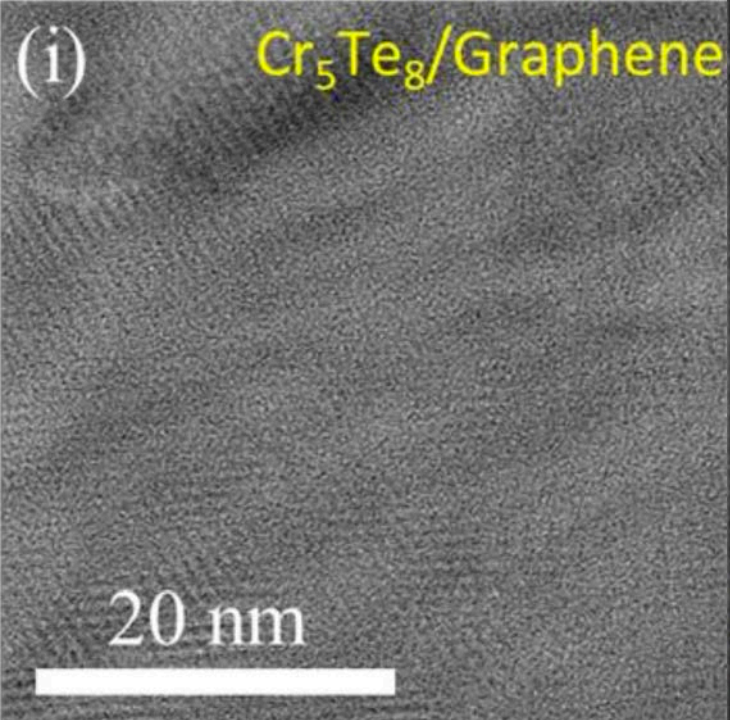
# Strain effect in vdW epitaxy

**Frank - van der Merwe growth**  
Layer-by-layer growth

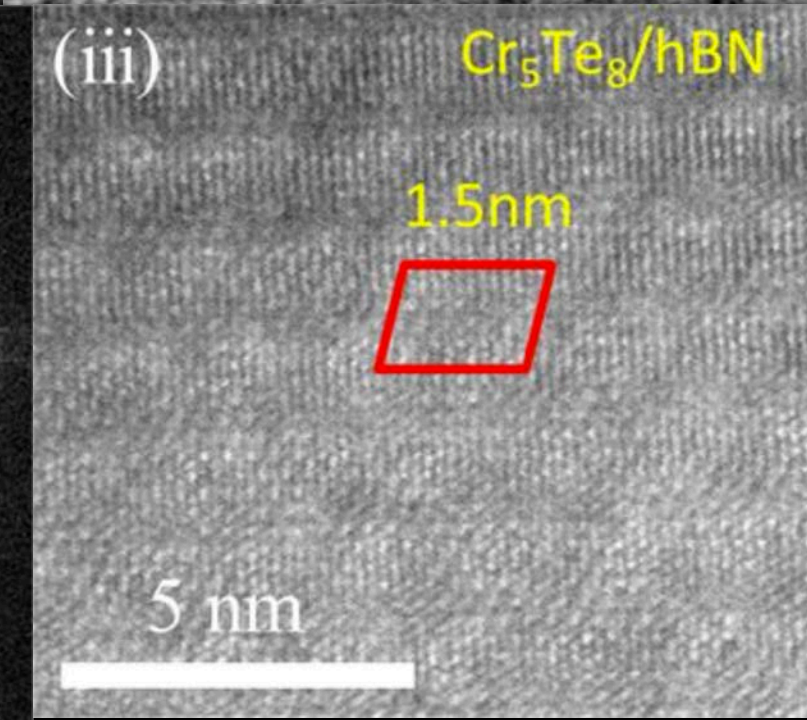
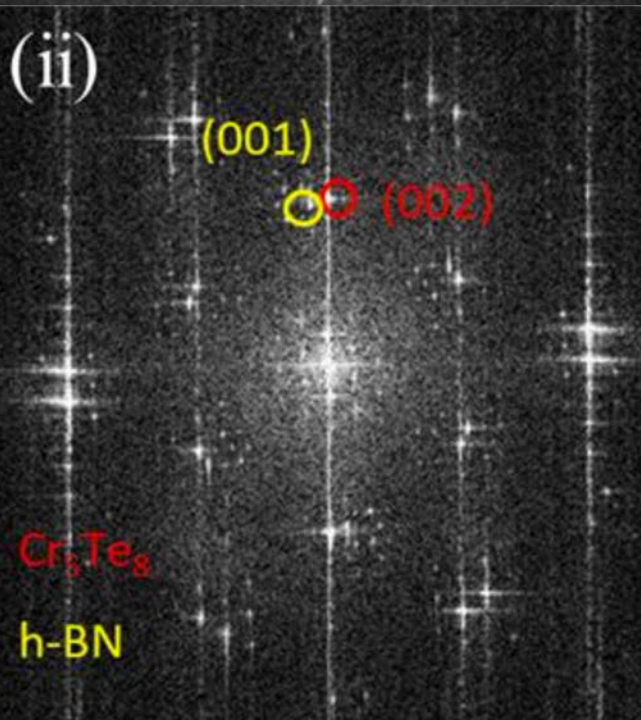
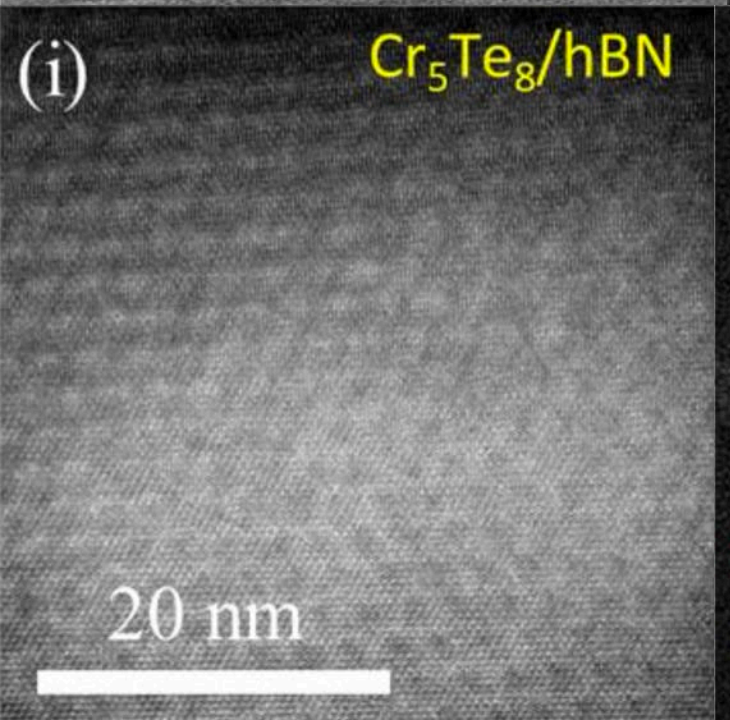


**Volmer - Weber growth**  
Island growth





$$\lambda = \frac{(1 + \delta)\alpha}{\sqrt{2(1 + \delta)(1 - \cos \phi) + \delta^2}}$$

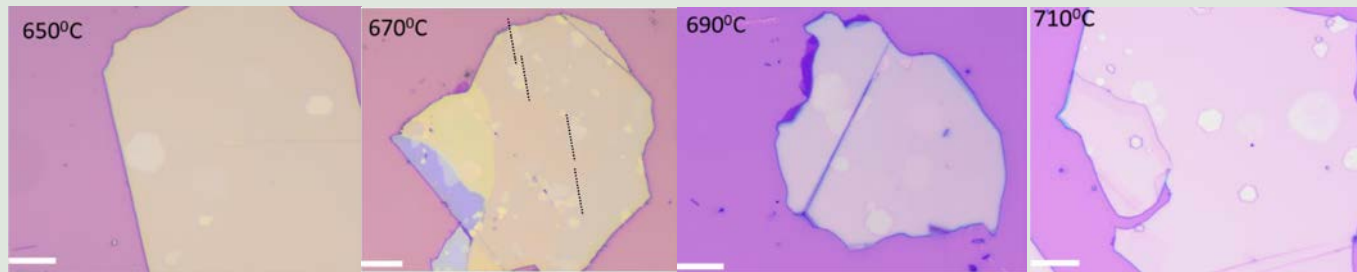


Moiré superlattice

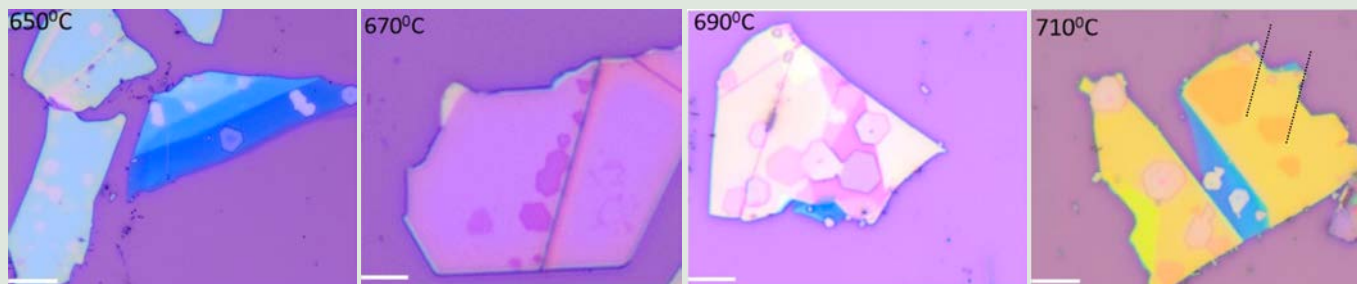
# Effect of growth parameters

## Temperature

## Precursors

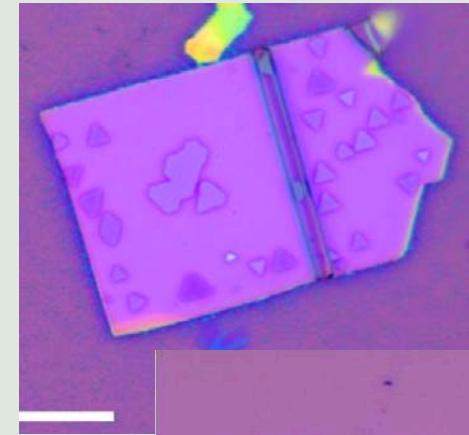


Graphite

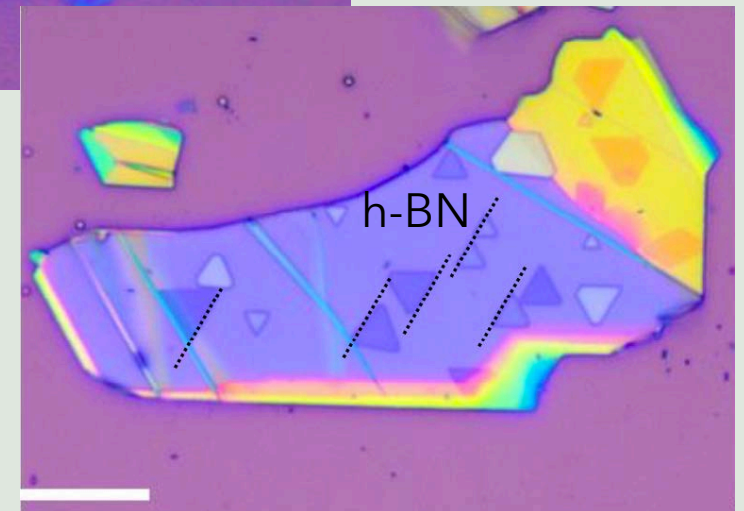


h-BN

Scale bar = 10  $\mu\text{m}$



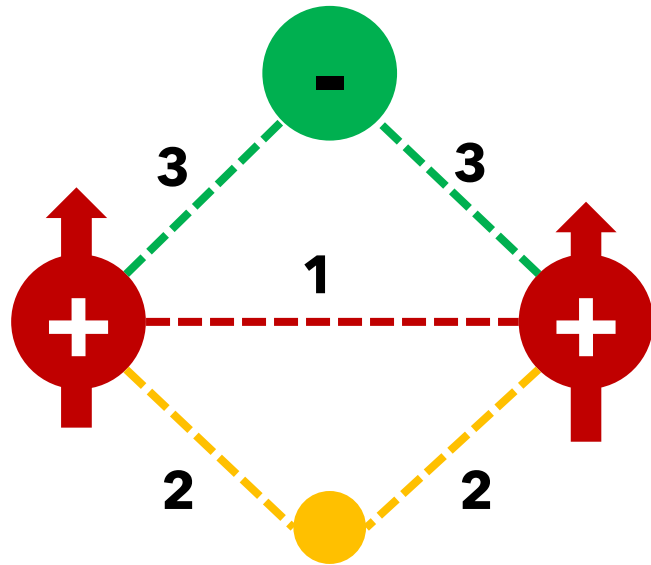
Salt-assisted CVD  
 $\text{CrCl}_3$ : NaCl (8:1) and Te



h-BN

# Magnetism in $\text{Cr}_{1+\delta}\text{Te}_2$

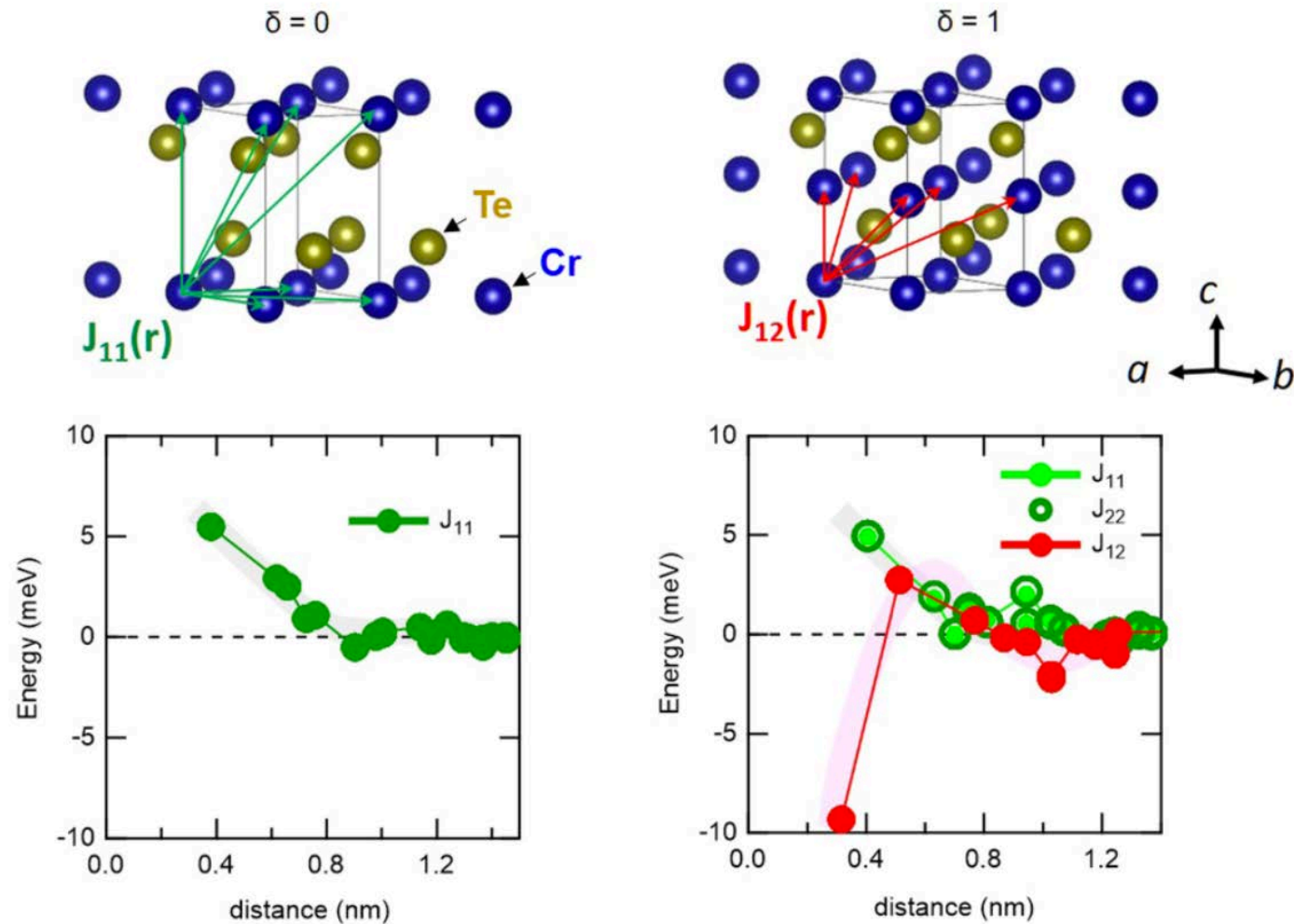
Exchange interaction



**1. Antiferromagnetic Cr-Cr direct exchange interaction**

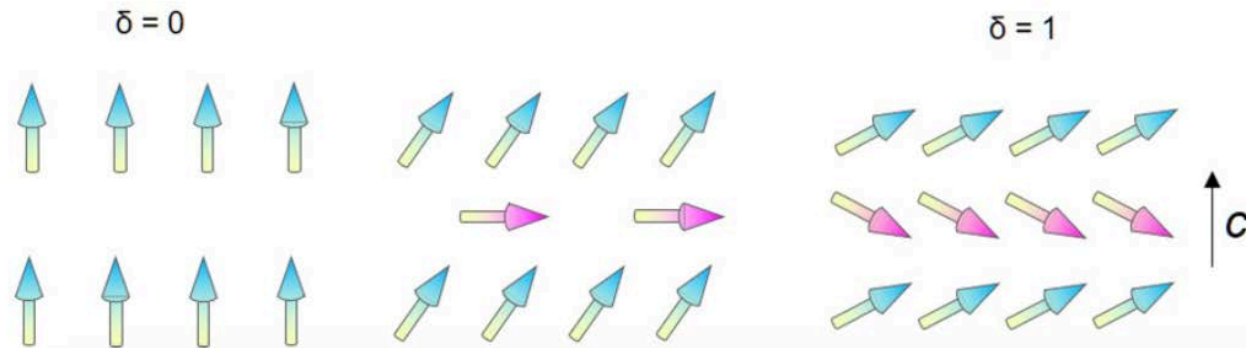
**2. Ferromagnetic  $t^3$ - $t^3$  superexchange interaction in the Cr-Te-Cr unit**

# Non-collinear Magnetism in $\text{Cr}_{1+\delta}\text{Te}_2$



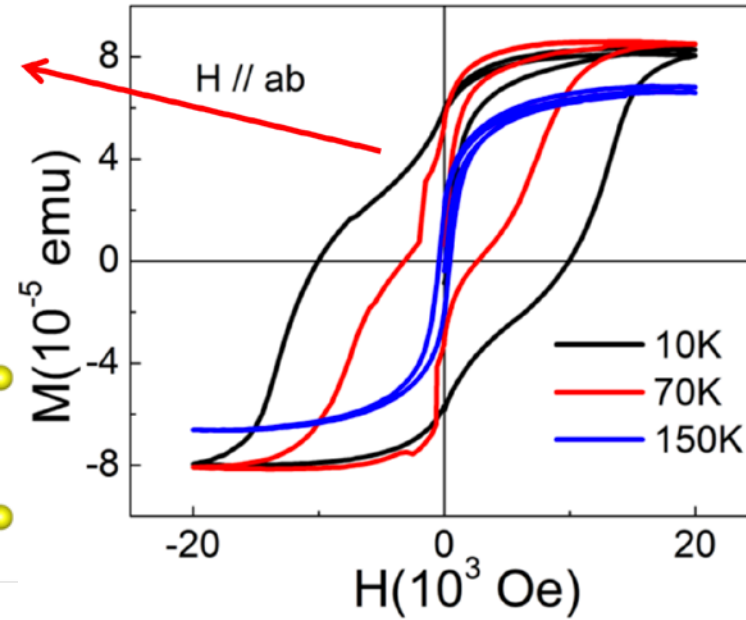
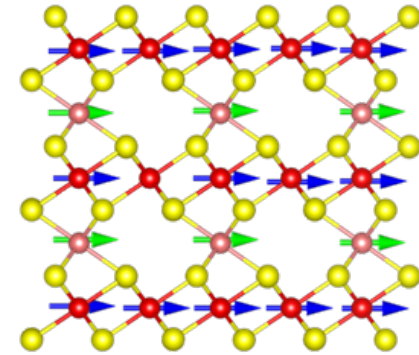
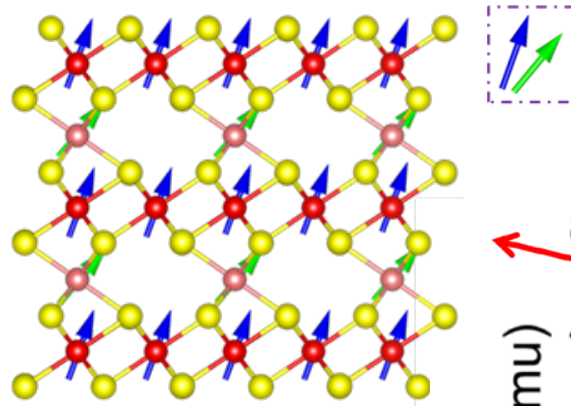
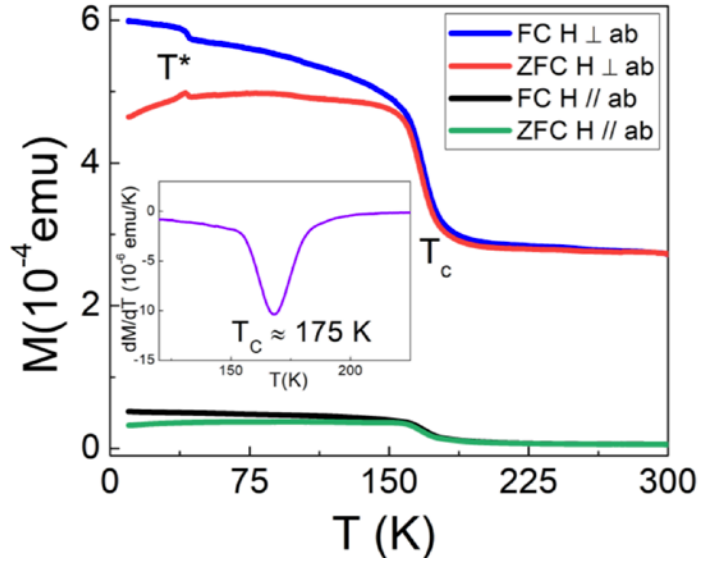
DFT-calculated direct exchange interactions for pairs of atoms intra-sublattice ( $J_{11}$ ,  $J_{22}$ ) and inter-sublattice ( $J_{12}$ ) as a function of atomic separation

Evolution of ground-state magnetic configuration of  $\text{Cr}_{1+\delta}\text{Te}_2$

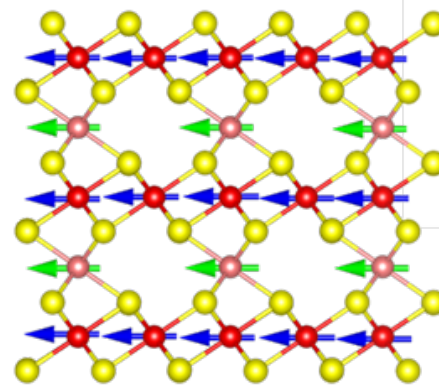
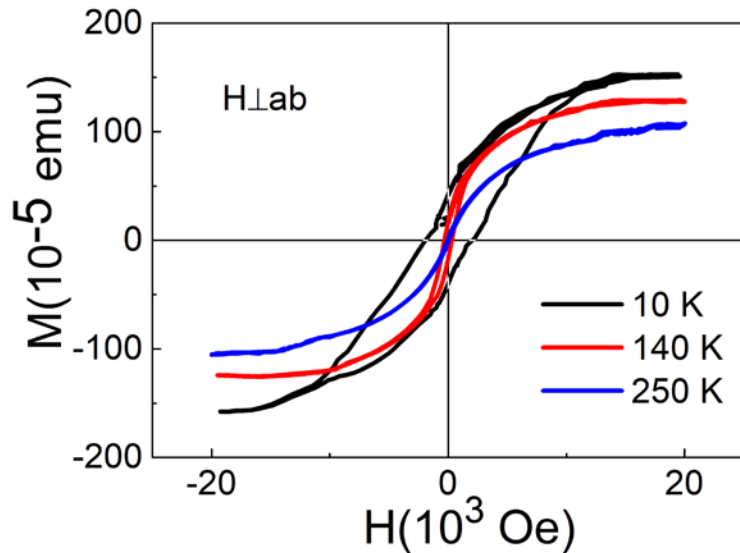




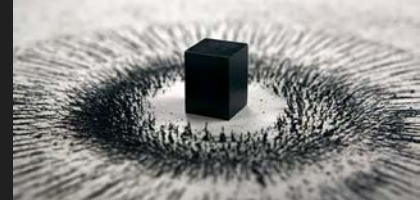
# Magnetism in $\text{Cr}_5\text{Te}_8$



- Cr I
- Cr II
- Te



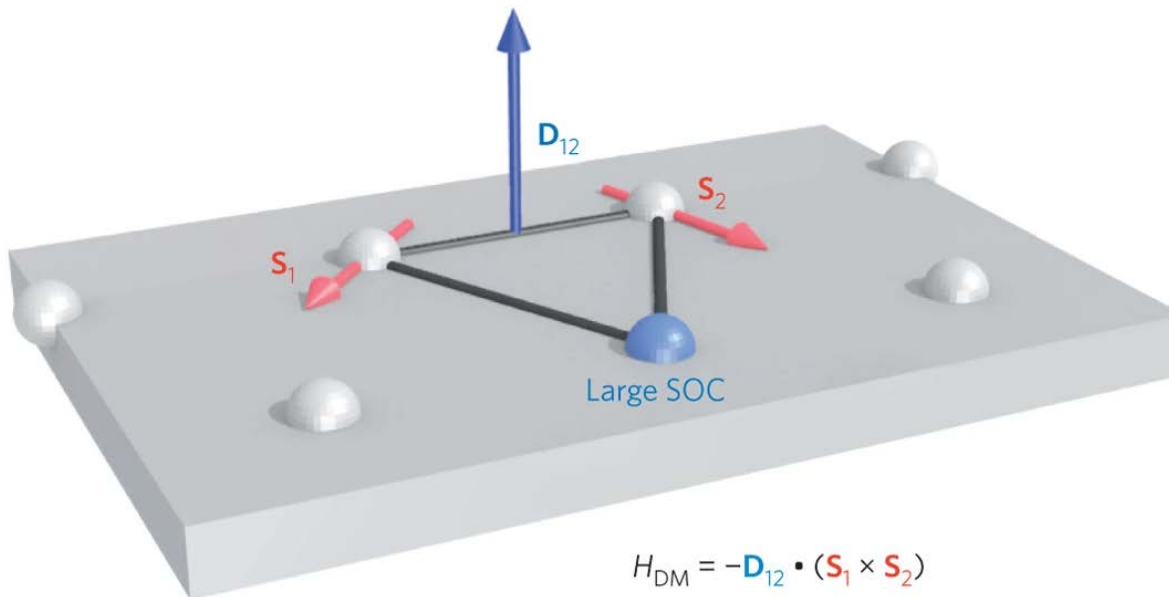
# Implications of non-collinear magnetism



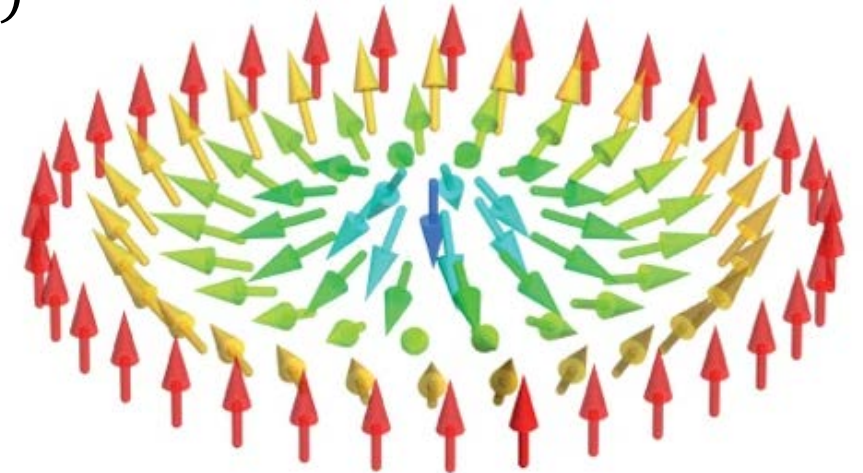
**Self-intercalated  $\text{Cr}_{1+\delta}\text{Te}_2$**  : Lacks inversion symmetry

**Dzyaloshinskii-Moriya interaction (DMI) / antisymmetric exchange** : Additional contribution to total magnetic exchange interaction in systems with broken inversion symmetry, when an atom with large spin-orbit coupling mediates a super-exchange interaction between two magnetic atoms.

$$\mathcal{H}_{DM} = -D_{12} \cdot (S_1 \times S_2)$$



$$H_{DM} = -D_{12} \cdot (S_1 \times S_2)$$

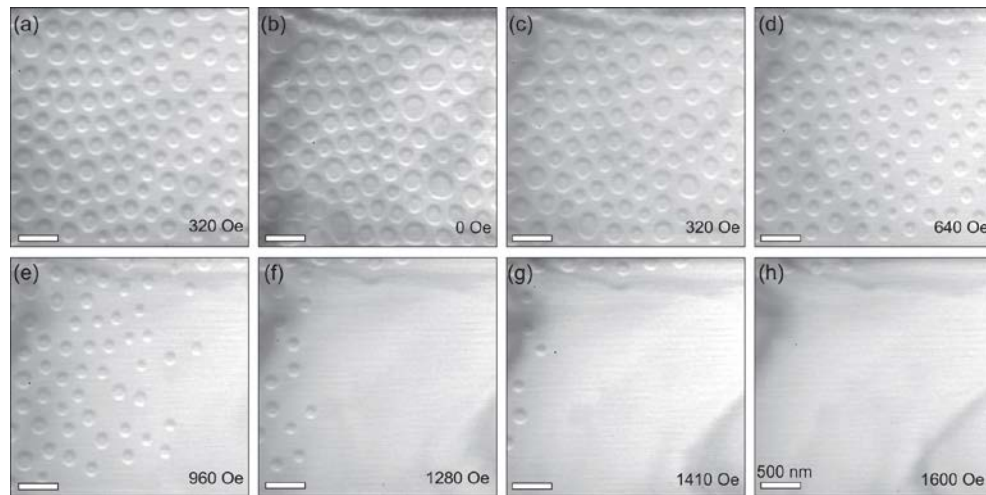




# Skyrmions in chromium tellurides

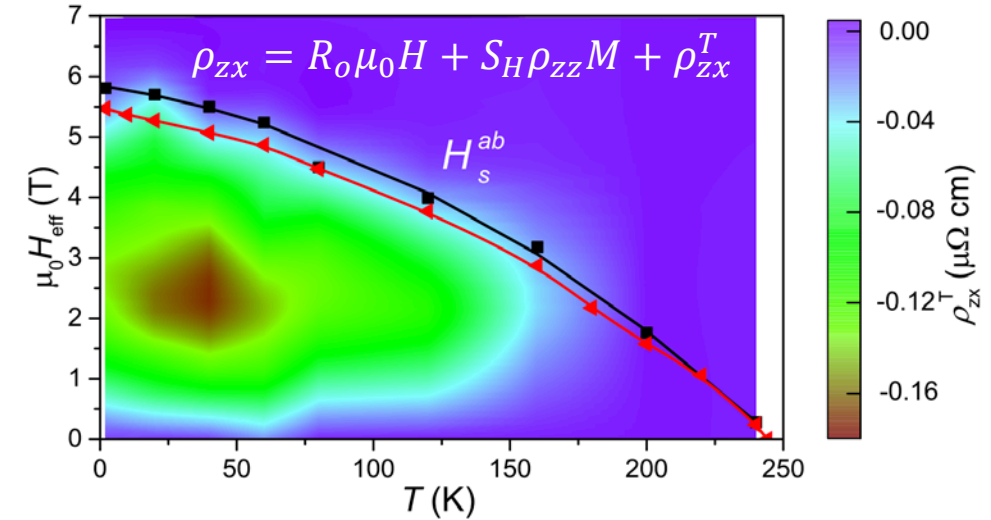
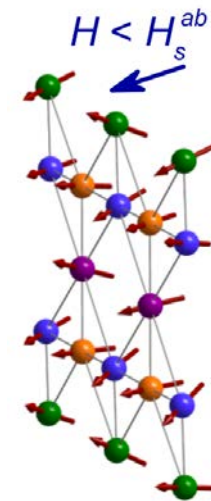


Evolution of Néel-type textures in  $Cr_{1+\delta}Te_2$  as a function of magnetic field at 100 K



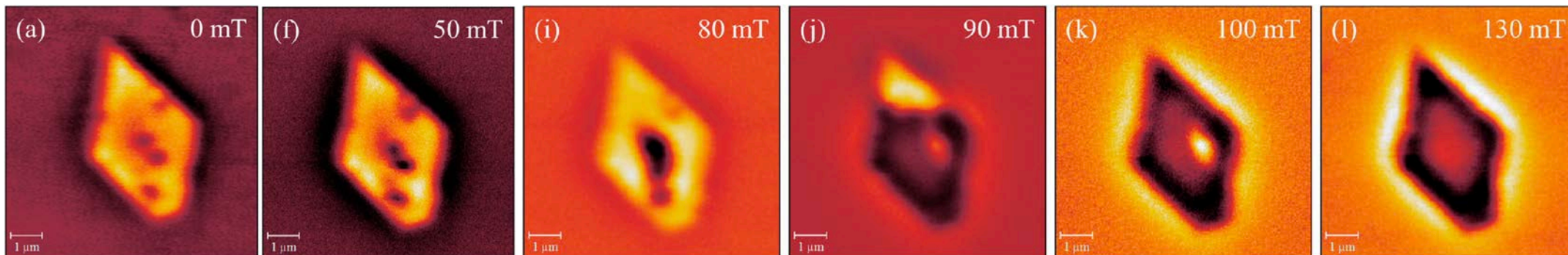
Nat Commun **13**, 3965 (2022)

Topological Hall effect in  $t-Cr_5Te_8$  and  $Cr_3Te_4$

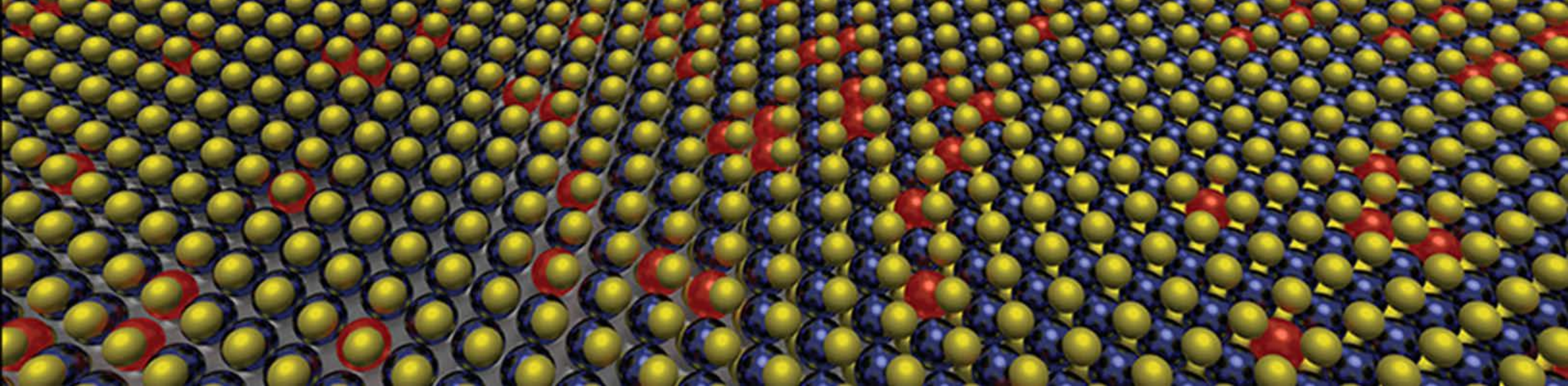


Phys. Rev. B **100**, 024434 (2019)  
Materials Today, 57, 66 (2022)

Evolution of magnetic bubbles in 2D  $Cr_5Te_8$



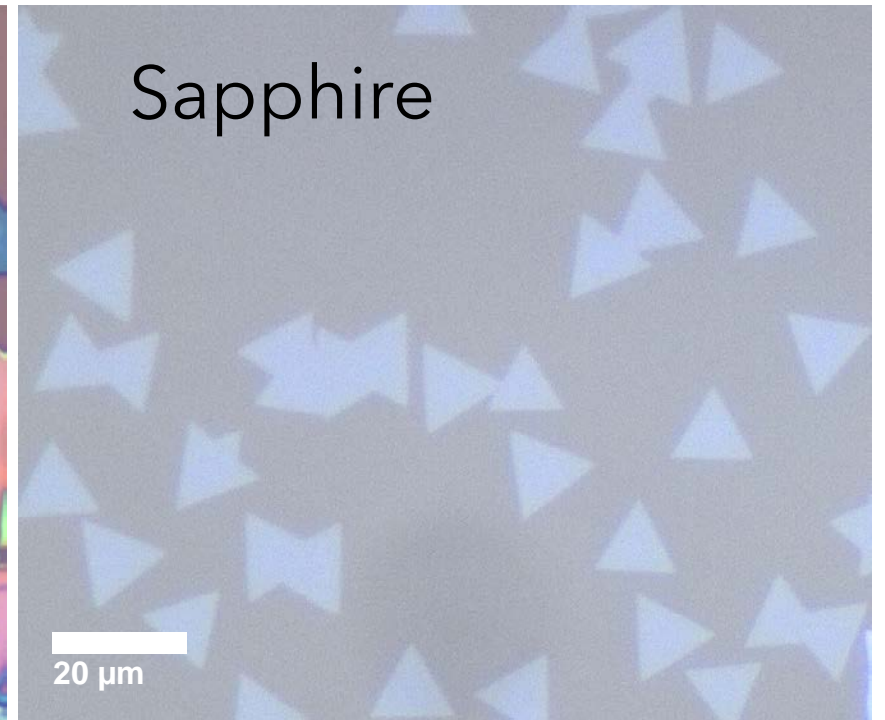
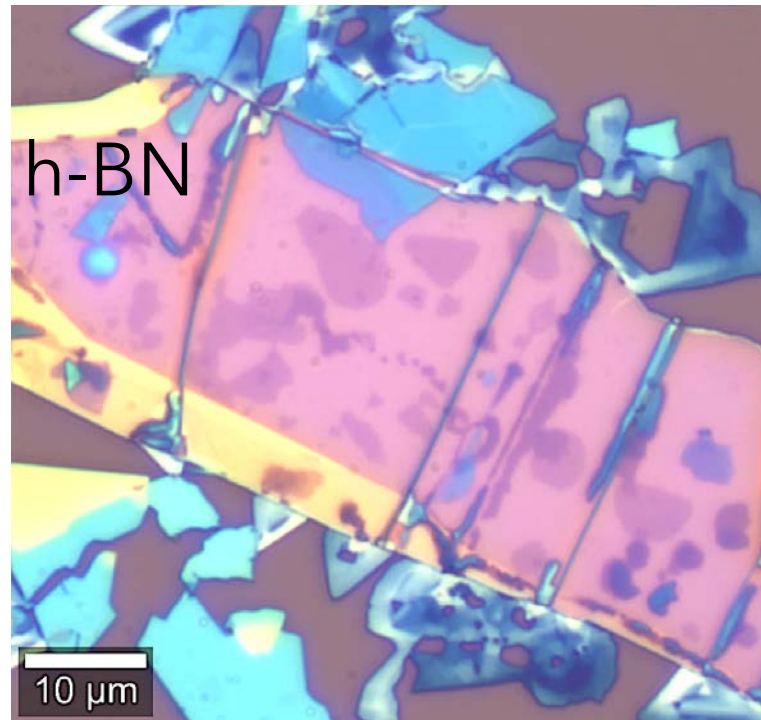
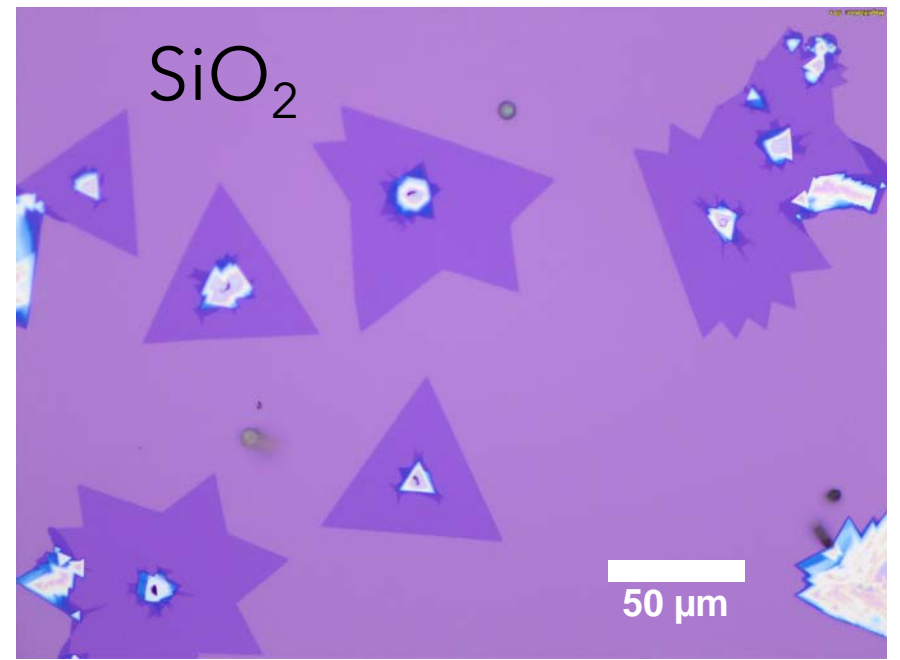
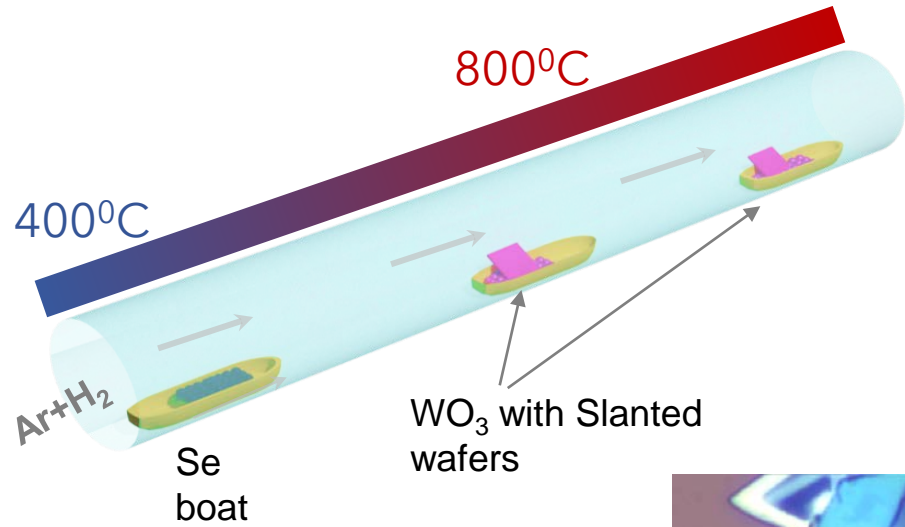
ACS Appl. Mater. Interfaces **15**, 26148 (2023)



## **2D semiconductors - WSe<sub>2</sub>**

Optical emission signatures and prospects  
for optoelectronics

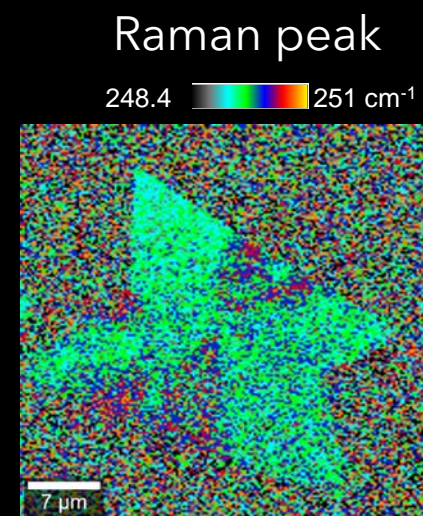
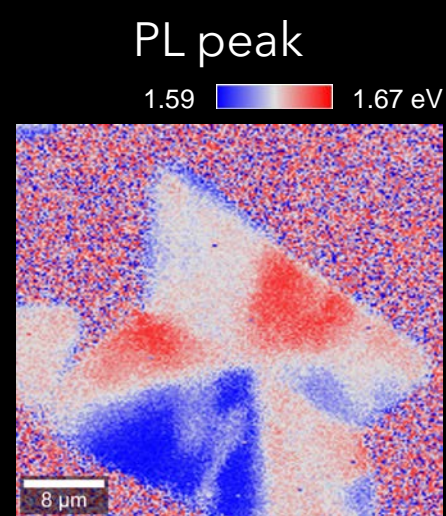
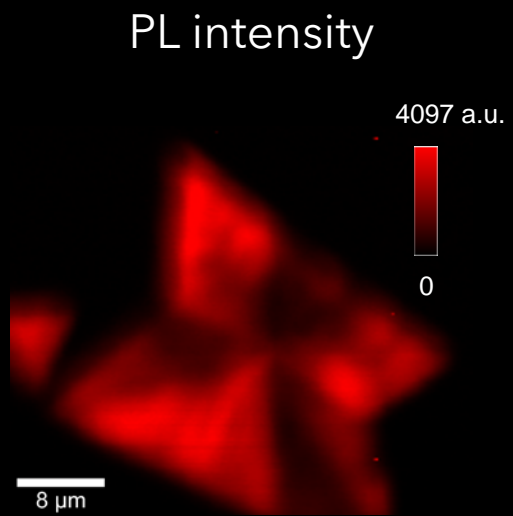
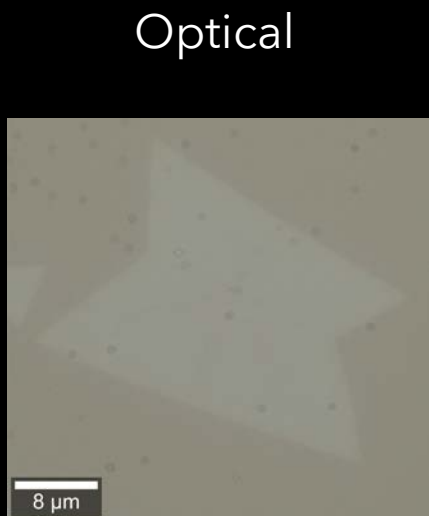
# CVD growth of WSe<sub>2</sub>



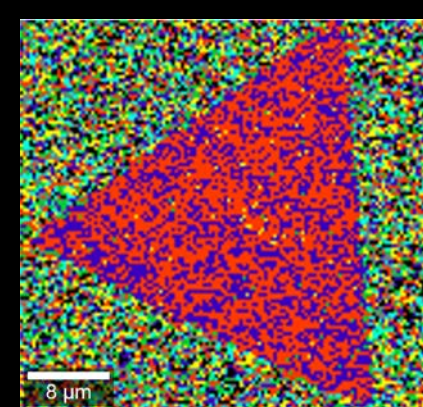
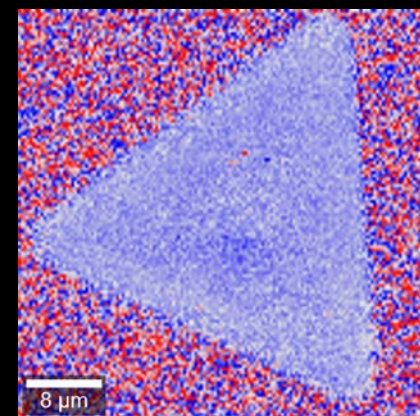
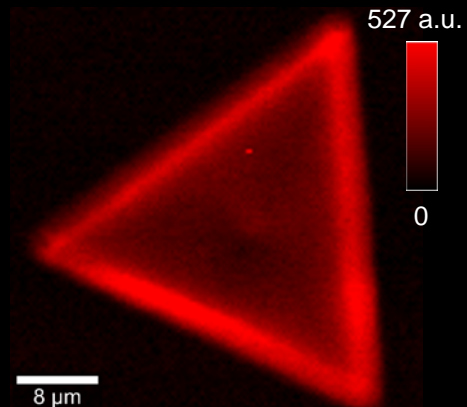
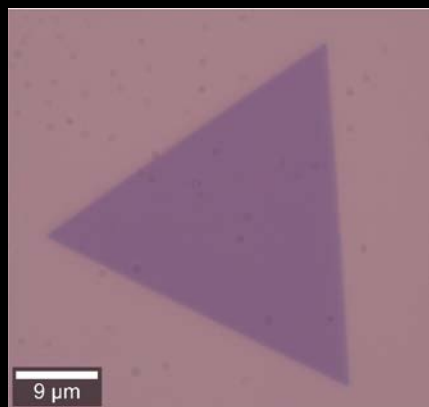
## Growth conditions

Temperature: 780- 800°C  
Gas Flow: 60 sccm Ar/ 8 sccm H<sub>2</sub>  
Growth Time: 5 min

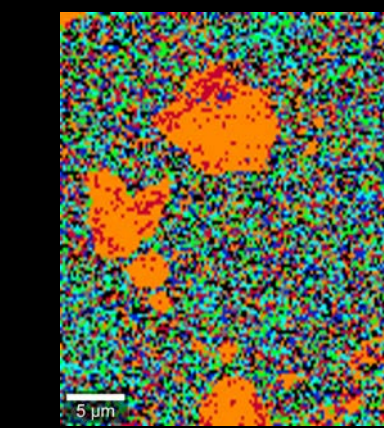
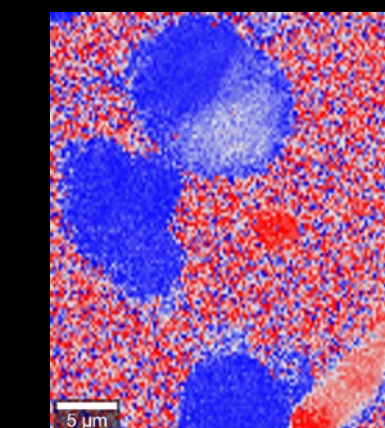
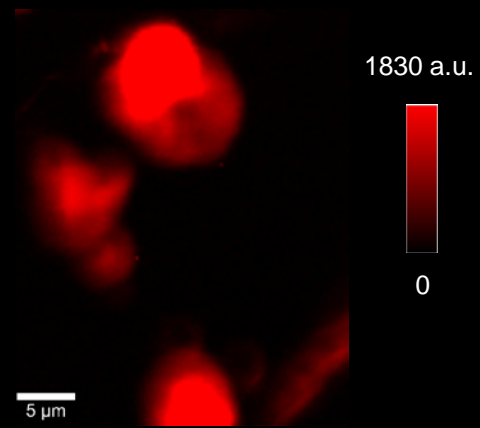
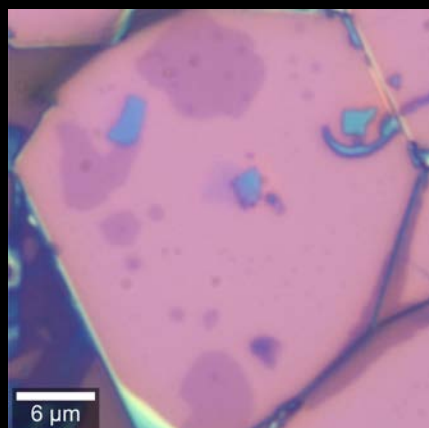
Sapphire



SiO<sub>2</sub>



h-BN



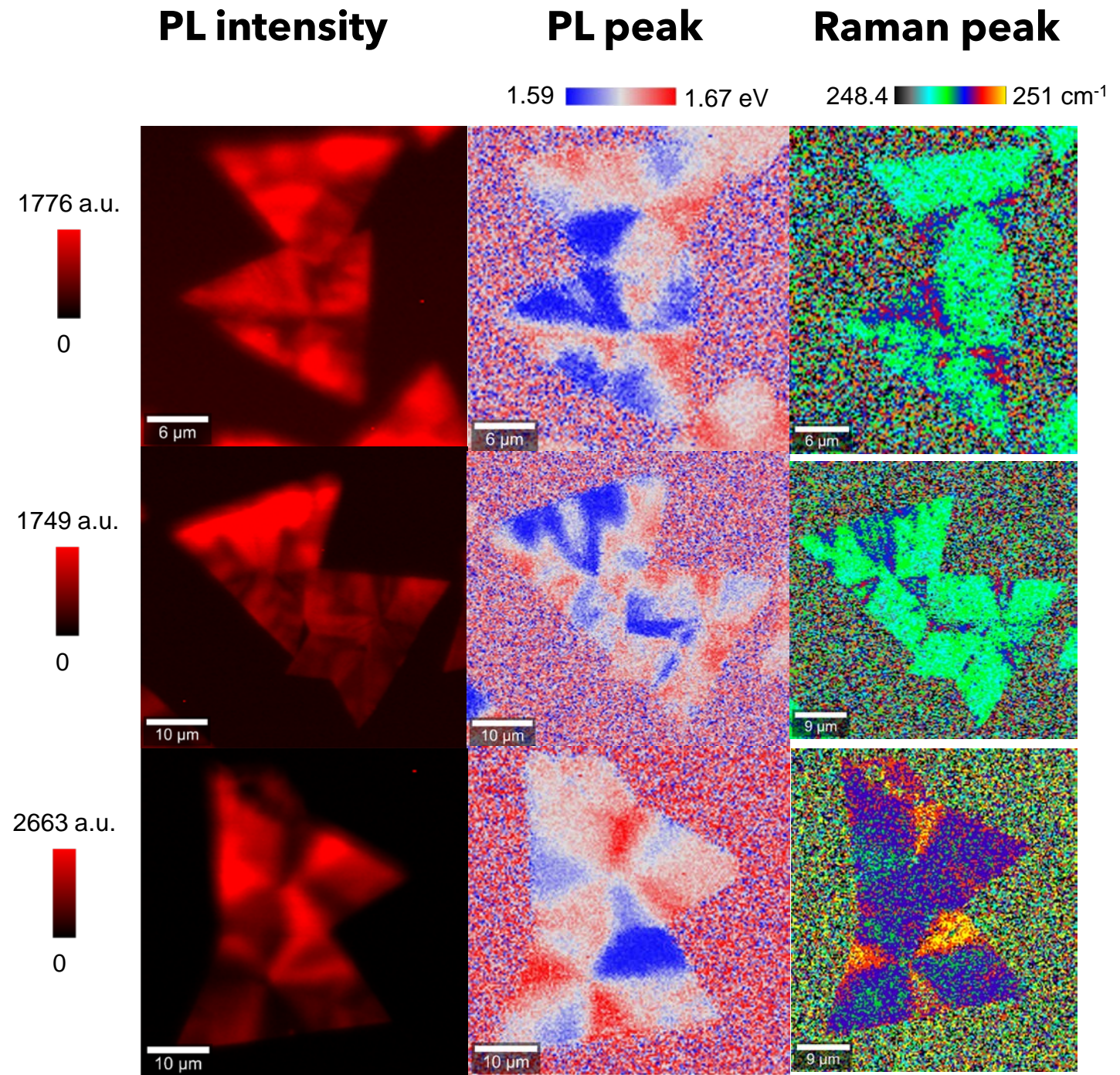
# TEC

Sapphire  
7.5 – 8.5  $\mu\text{/K}$

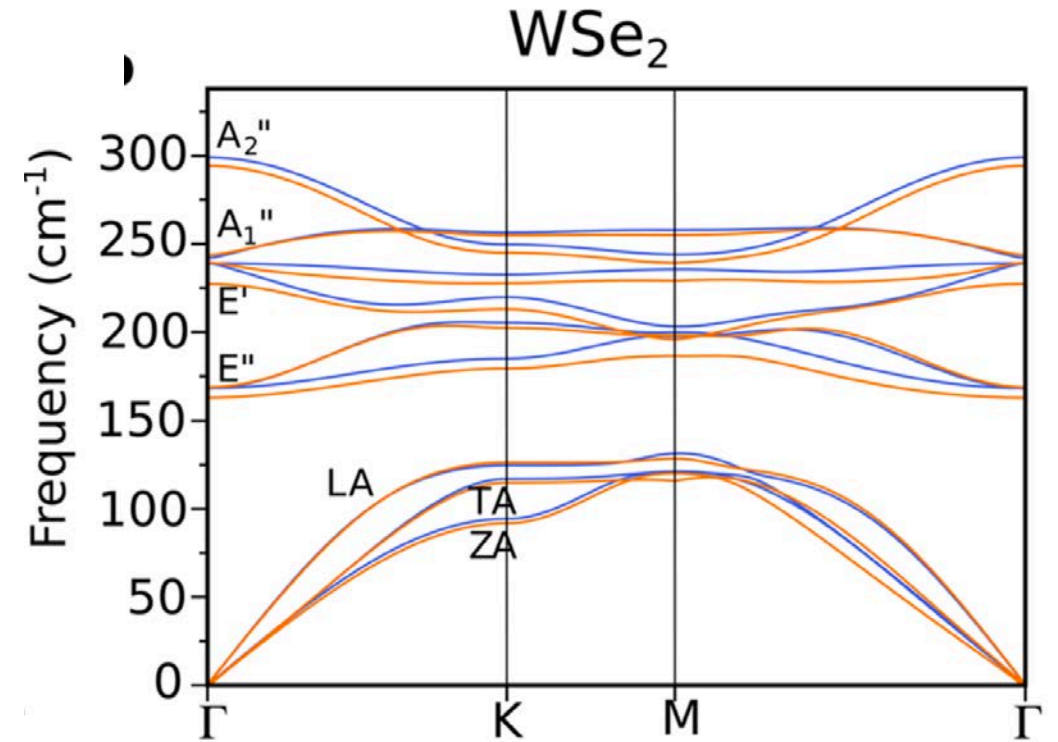
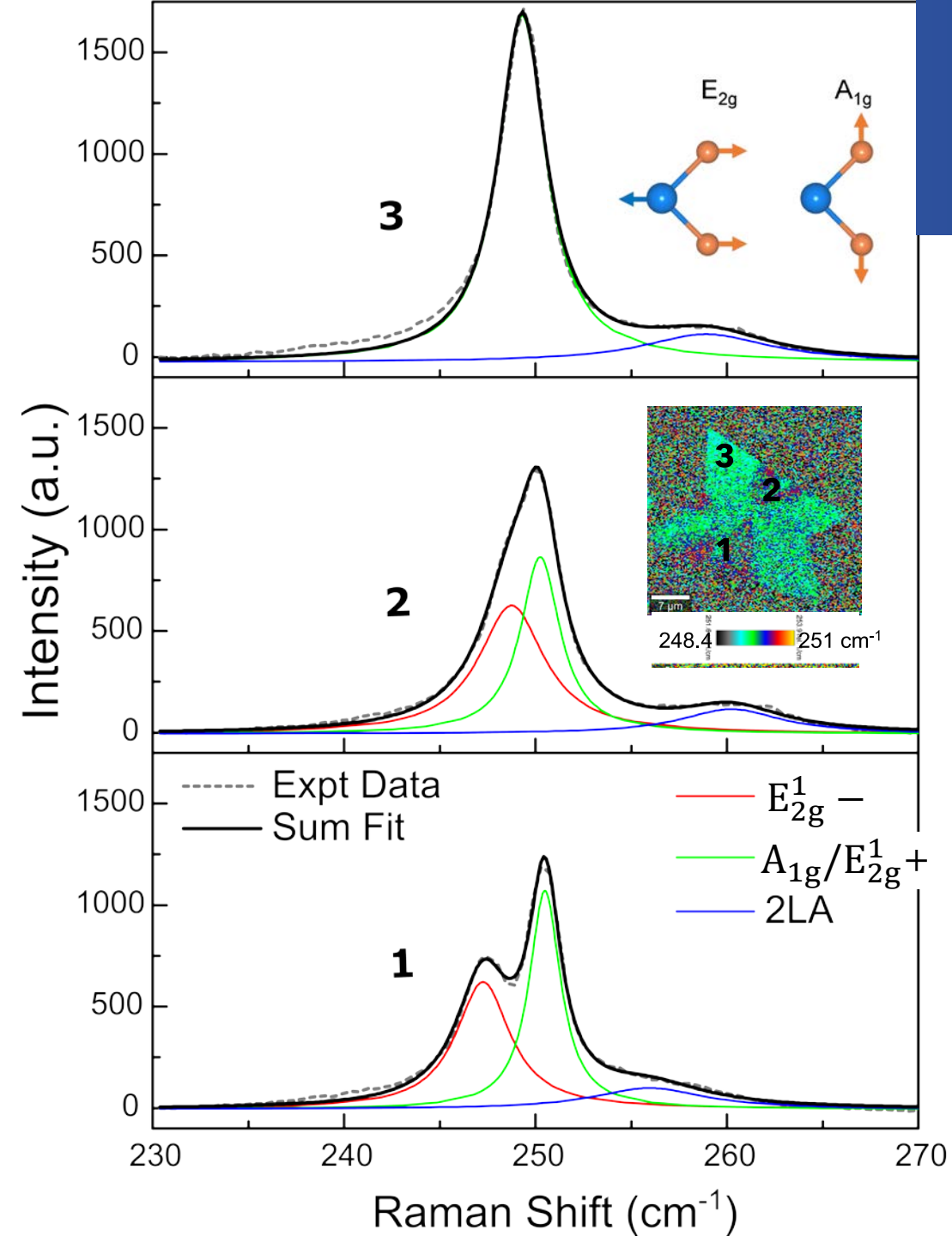
h-BN  
-2.72  $\mu\text{/K}$

SiO<sub>2</sub>  
0.5  $\mu\text{/K}$

WSe<sub>2</sub>  
3.5  $\mu\text{/K}$

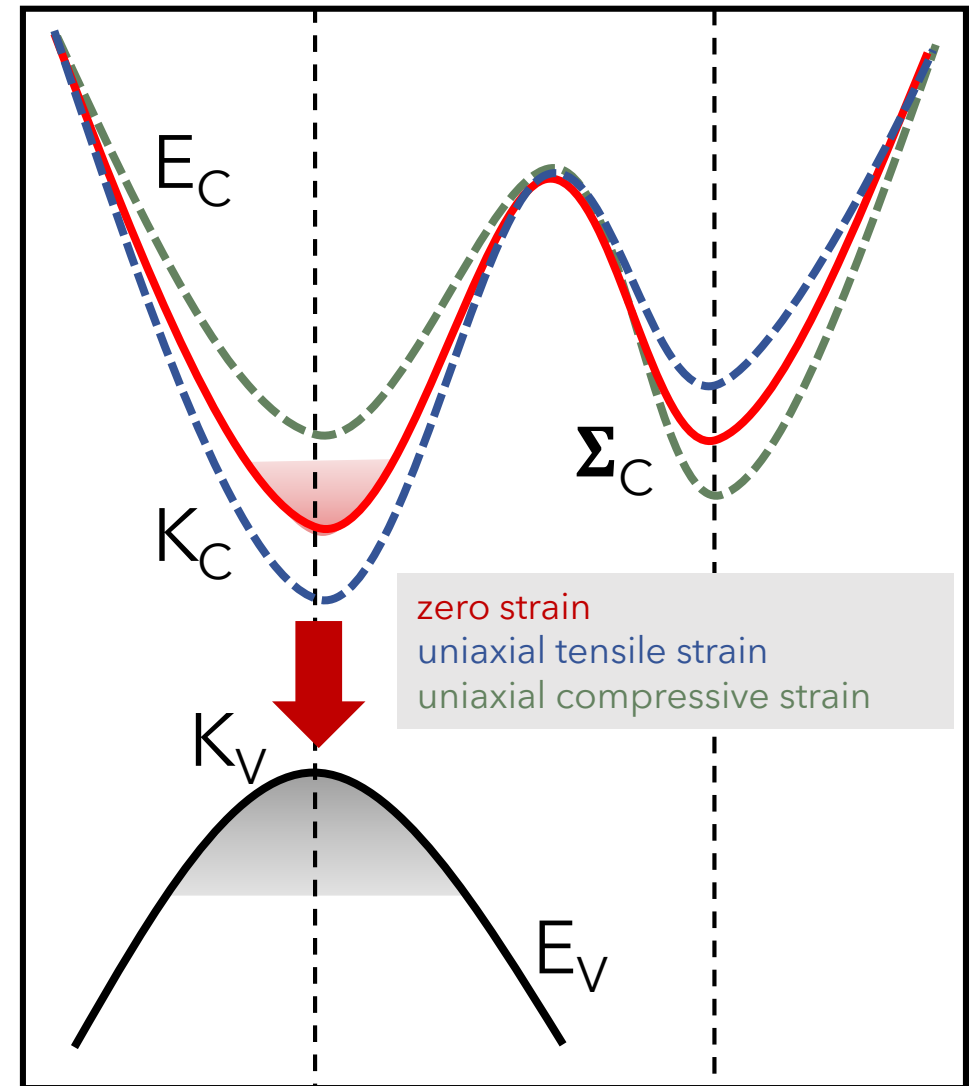
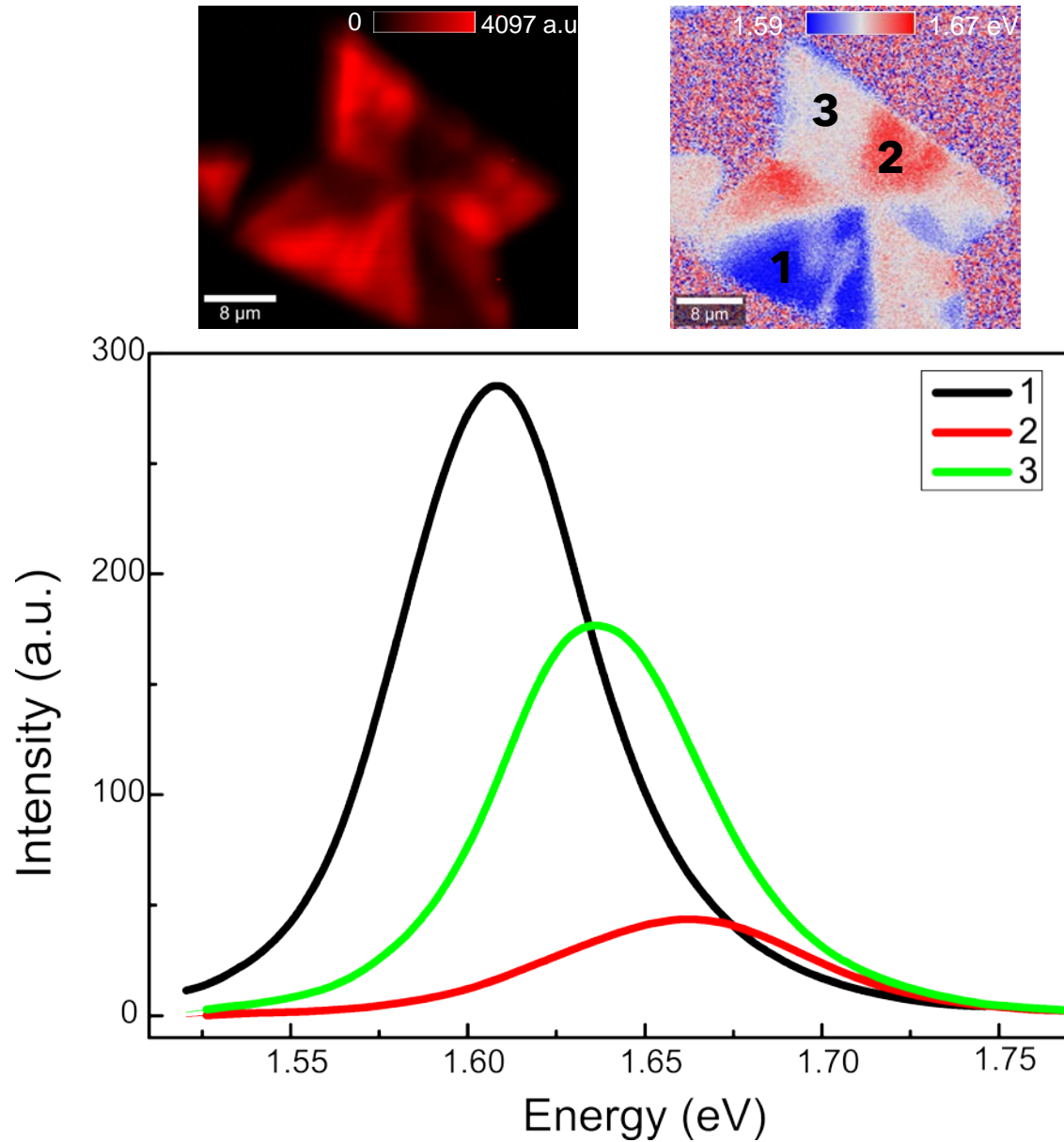


# Substrate strain effect on Raman spectra



The displacement eigenvector of the E'+ mode is orthogonal to the direction of strain, while it is parallel for the E'- mode .

# Substrate strain effect on Photolumuminescence





# Conclusions and future outlook

- Van der Waals epitaxy - Growth mechanism, prospects and challenges for large area growth of 2D/quasi-2D materials
- $\text{Cr}_{1+x}\text{Te}_2$  system - non-collinear magnetism and prospects for hosting skyrmions
- Substrate interactions in  $\text{WSe}_2$  - effect of strain on vibrational modes and band structure



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IIT Kharagpur

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Department of Science & Technology

Government of India



Thanks for your kind attention...